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Part 7: Insulations

National foreword

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Table of contents

European Foreword.....	14
1 Scope.....	15
2 References	16
3 Terms, definitions and symbols	17
3.1 Terms and definitions	17
3.2 Abbreviated terms.....	17
3.3 Symbols.....	17
4 Foams	22
4.1 General.....	22
4.2 Inorganic foams	25
4.3 Organic foams	31
4.3.2 Thermal properties of organic foams	32
4.3.3 Mechanical properties of organic foams	35
4.3.4 Data on commercially available foams	53
5 Fibrous insulations	64
5.1 General.....	64
5.2 Bults.....	66
5.3 Blankets and felts	77
5.4 Papers	102
6 Multilayer insulations	108
6.1 General.....	108
6.1.1 Fundamental concepts concerning MLI performance	109
6.1.2 Failure modes	111
6.1.3 Heat transfer through an MLI	111
6.1.4 Cost	118
6.2 Radiation shields	118
6.2.1 Aluminium foils and aluminium coated plastic films	118
6.2.2 Gold foils and gold coated plastic films.....	119
6.2.3 Silver coated plastic films	119

6.2.4	Operating temperature ranges	119
6.2.5	Normally used plastic films	120
6.3	Emittance of metallic foils	120
6.4	Emittance of metallized films	135
6.5	Absorptance of metallic foils	148
6.6	Radiation shields. miscellaneous properties	165
6.7	Radiation shields. measurement of the coating thickness	180
6.8	Spacers	183
6.8.1	Multiple-resistance spacers	184
6.8.2	Point-contact spacers	184
6.8.3	Superfloc	184
6.8.4	Single-component MLI	185
6.8.5	Composite spacers	185
6.8.6	One-dimensional heat flow through an mli with absorbing and scattering spacers	187
6.9	Spacers. miscellaneous properties	189
6.10	Complete systems	208
6.11	Normal heat transfer	209
6.12	Lateral heat transfer	251
6.13	Effect of singularities	257
6.13.1	Joints	257
6.13.2	Stitches and patches	269
6.14	Effect of evacuating holes	272
6.15	Effect of mechanical damage	276
6.16	Effect of inner gas pressure	277
6.17	Evacuation	285
6.17.1	Interstitial pressure during rapid evacuation	285
6.17.2	Interstitial pressure in outgas controlled situations	292
6.17.3	Self-pumping multilayer insulations	301
Bibliography		319

Figures

Figure 4-1: Resin thermal conductivity	23
Figure 4-2 : Gas thermal conductivity	23
Figure 4-3: Radiation thermal conductivity	24
Figure 4-4: Thermal conductivity, k , of several ceramic foams as a function of arithmetic mean temperature, T_m	26

Figure 4-5: Linear thermal expansion, $\Delta L/L$, of several ceramic foams as a function of temperature, T	27
Figure 4-6: Temperature evolution of the hot and cold faces of several pieces of Zircon foam. Solid line: T_H , hot face. Dashed line: T_C , cold face.	28
Figure 4-7: Thermal conductivity k , of polyurethane foams vs. arithmetic mean temperature, T_m	32
Figure 4-8: Thermal conductivity, k , of cryopumped polystyrene foams.....	33
Figure 4-9: Thermal conductivity, k , vs. arithmetic mean temperature, T_m , of a polyurethane foam in the proximity of the condensation temperature of the filling gas.	34
Figure 4-10: Linear thermal expansion, $\Delta L/L$, of several organic foams as a function of temperature, T	35
Figure 4-11: Ultimate tensile strength, of several foams as a function of temperature, T	36
Figure 4-12: Ultimate shear strength, τ_{ult} , of several foams as a function of temperature, T	37
Figure 4-13: Tensile stress, σ , vs. strain, δ , for several polyurethane foams at 76, 195 and 300 K.....	46
Figure 4-14: Modulus of Elasticity-tensile- E , as a function of density, ρ , for several organic foams.....	46
Figure 4-15: Ultimate tensile strength, σ_{ult} , as a function of density, ρ , for several organic foams.....	47
Figure 4-16: Compressive stress, s , vs. strain, d , for several organic foams at 76, 195 and 300 K.....	47
Figure 4-17: Modulus of Elasticity-tensile- E , as a function of density, ρ , for several organic foams.....	48
Figure 4-18: Proportional limit-compressive- σ , as a function of density, ρ , for several organic foams.....	49
Figure 4-19: Ultimate tensile strength, σ_{ult} , and Modulus of Elasticity-tensile- E , as functions of temperature, T	50
Figure 4-20: Ultimate compressive strength, σ_{ult} , and Modulus of Elasticity-compressive- E , as functions of temperature, T	51
Figure 4-21: Ultimate compressive strength, σ_{ult} , as a function of temperature, T	52
Figure 4-22: Ultimate block shear strength, τ_{ult} , and Modulus of Elasticity-shear block- E , as functions of temperature, T	53
Figure 4-23: Strain, δ , vs. compressive stress, σ , of Fiberfill Structural Foams.	62
Figure 4-24: Dielectric constant, ϵ_r , and dissipation factor, D , vs. frequency, f . Stycast 1090.....	62
Figure 5-1: Thermal conductivity, k , vs. mean temperature, T_m , for several fibrous insulations. From Glasser et al. (1967) [23].	65
Figure 5-2: Thermal conductivity, k , of B & W Kaowool bulk vs. mean temperature, T_m	74
Figure 5-3: Thermal conductivity, k , of Carborundum Fiberfrax bulk and washed fibers vs. mean temperature, T_m	74
Figure 5-4: Temperature differential, $T_H - T_C$, vs. mean temperature of the hot face, T_H ,.....	75

Figure 5-5: Temperature differential, $T_H - T_C$, vs. mean temperature of the hot face, T_H ,.....	75
Figure 5-6: Temperature differential, $T_H - T_C$, vs. mean temperature of the hot face, T_H ,.....	76
Figure 5-7: Temperature differential, $T_H - T_C$, vs. mean temperature of the hot face, T_H for different values of the insulation thickness, t . Fiberfrax washed fiber, $\rho =$ 96 kg.m^{-3}	76
Figure 5-8: Sound absorption coefficient, α , as a function of frequency, f , for B & W Kaowool blanket $2,54 \times 10^{-2} \text{ m}$ thick.....	86
Figure 5-9: Sound absorption coefficient, α , as a function of frequency, f , for the following Fiberfrax products:.....	86
Figure 5-10: Air permeability across B & W Kaowool blankets.....	87
Figure 5-11: Air permeability across Carborundum Fiberfrax blankets.....	88
Figure 5-12: Thermal conductivity, k , of B & Kaowool blankets vs. mean temperature, T_m	88
Figure 5-13: Thermal conductivity, k , of Fiberfrax blankets vs. mean temperature, T_m	89
Figure 5-14: Temperature differential, $T_H - T_C$, vs. temperature of the hot face T_H for different values of the blanket thickness, t . Fiberfrax Lo-Con Blanket & felt, $\rho = 64 \text{ kg.m}^{-3}$	89
Figure 5-15: Temperature differential, $T_H - T_C$, vs. temperature of the hot face, T_H ,	90
Figure 5-16: Sound absorption coefficient, α , as a function of frequency, f , for J-M Microlite Standard and Silicone Binder.	91
Figure 5-17: Calculated specific heat, c , as a function of temperature, T , for J.M Dyna- Quartz	91
Figure 5-18: Thermal conductivity, k , of J-M Micro-Quartz felt vs. mean temperature, T_m	93
Figure 5-19: Thermal conductivity, k , of J-M Dyna-Quartz vs. mean temperature, T_m	93
Figure 5-20: Thermal conductivity, k , of J-M Microlite "AA" and "B" vs. mean temperature, T_m	94
Figure 5-21: Compressive stress, σ , vs. compressive strain, δ , for J-M Dyna-Quartz.	94
Figure 5-22: Linear Shrinkage, LS , and Total Weight Loss, TWL , of J-M Micro-Quartz Felt as a function of temperature, T	95
Figure 5-23: Calculated specific heat, c , as a function of temperature, T , for several J-M insulations.	96
Figure 5-24: Thermal conductivity, k , of J-M Min-K 1301 vs. mean temperature, T_m . Numbers on curves indicate the density in kg.m^{-3}	97
Figure 5-25: Influence of ambient pressure on the variation of thermal conductivity, k , of J-M Min-K 1301 vs. mean temperature, T_m , for several filling gases.....	97
Figure 5-26: Thermal conductivity, k , of J-M Min-K 1301 vs. mean temperature, T_m , for different values of thickness, t	98
Figure 5-27: Thermal conductivity, k , of J-M Min-K 2000 vs. mean temperature, T_m	98
Figure 5-28: Influence of ambient pressure on the variation of thermal conductivity, k , of J-M Min-K 2000 vs. mean temperature, T_m , for several filling gases.....	99
Figure 5-29: Thermal conductivity, k , of J-M Min-K 2000 vs. mean temperature, T_m , for different values of thickness, t	99

Figure 5-30: Thermal conductivity, k , of J-M unbounded B-Fiber batt vs. mean temperature, T_m	100
Figure 5-31: Thermal conductivity, k , of J-M Micro-Fibers felt Type "E" vs. mean temperature, T_m	100
Figure 5-32: Compressive stress, σ , vs. compressive strain, δ , for J-M Min-K 1301. Numbers on curves indicate the density in kg.m^{-3}	101
Figure 5-33: Compressive stress, σ , vs. compressive strain, δ , for J-M Min-K 2000.	101
Figure 5-34: Thermal conductivity, k , of B & W Kaowool, Carborundum Fiberfrax 970 paper, and Fiberfrax Hi-Fi 660 paper vs. mean temperature, T_m	106
Figure 5-35: Temperature differential, $T_H - T_C$, vs. temperature of the hot face, T_H , for different values of the paper thickness, t . Fiberfrax 970 paper, $\rho = 160 \text{ kg.m}^{-3}$	106
Figure 5-36: Temperature differential, $T_H - T_C$, vs. temperature of the hot face, T_H , for different values of the paper thickness, t . Fiberfrax Hi-Fi 660 paper.	107
Figure 5-37: Thermal reflectance, ρ , of Fiberfrax 970-J paper vs. mean temperature, T_m	107
Figure 6-1: Effective thermal conductivity, k_{eff} , of multilayer insulations as compared with other insulation materials. From Glaser et al. (1967) [23].	109
Figure 6-2: Effective thermal conductivity, k_{eff} , of several multilayer insulation systems as a function of the characteristic temperature, T . Calculated by the compiler.	116
Figure 6-3: Summary of data concerning hemispherical total emittance, ε , of Aluminium foils and thin sheets as a function of temperature, T	121
Figure 6-4: Summary of data concerning hemispherical total emittance, ε , of Copper as a function of temperature, T	122
Figure 6-5: Hemispherical total emittance, ε , of Copper as a function of temperature, T	123
Figure 6-6: Hemispherical total emittance, ε , of Copper as a function of temperature, T	124
Figure 6-7: Hemispherical total emittance, ε , of Gold vs. temperature, T	125
Figure 6-8: Hemispherical total emittance, ε , of Molybdenum vs. temperature, T	126
Figure 6-9: Hemispherical total emittance, ε , of Nickel vs. temperature, T	128
Figure 6-10: Hemispherical total emittance, ε , of oxidized Nickel as a function of temperature, T	129
Figure 6-11: Normal total emittance, ε' , of Nickel as a function of temperature, T	130
Figure 6-12: Normal total emittance, ε' , of Inconel as a function of temperature, T	131
Figure 6-13: Normal total emittance, ε' , of Inconel X as a function of temperature, T	132
Figure 6-14: Hemispherical total emittance, ε , of Platinum as a function of temperature, T	133
Figure 6-15: Hemispherical total emittance, ε , of Silver as a function of temperature, T	134
Figure 6-16: Hemispherical total emittance, ε , of Aluminized Mylar as a function of coating thickness, t_c	136
Figure 6-17: Hemispherical total emittance, ε , of Copper on Mylar as a function of coating thickness, t_c	137

Figure 6-18: Hemispherical total emittance, ε , of Goldized Mylar as a function of coating thickness, t_c	138
Figure 6-19: Hemispherical total emittance, ε , of Gold on Double Aluminized Mylar as a function of Gold thickness, t_c	139
Figure 6-20: Hemispherical total emittance, ε , of Silvered Mylar as a function of coating thickness, t_c	140
Figure 6-21: Hemispherical total emittance, ε , of Silvered Mylar overcoated with Silicon Monoxide as a function of Silver thickness, t_c	141
Figure 6-22: Hemispherical total emittance, ε , of Aluminized Kapton as a function of temperature, T	143
Figure 6-23: Hemispherical total emittance, ε , of Aluminized Kapton as a function of coating thickness, t_c , for $T = 300$ K.	144
Figure 6-24: Hemispherical total emittance, ε , of Aluminized Kapton as a function of coating thickness, t_c , for $T = 400$ K.	145
Figure 6-25: Hemispherical total emittance, ε , of Silgle-Goldized Kapton as a function of temperature, T	146
Figure 6-26: Hemispherical total emittance, ε , of Silgle-Silvered Kapton as a function of temperature, T	147
Figure 6-27: Normal spectral absorptance, α'_λ , of Aluminium as a function of wavelength, λ	148
Figure 6-28: Normal solar absorptance, α_s , of Aluminium as a function of temperature, T	149
Figure 6-29: Normal spectral absorptance, α'_λ , of Copper as a function of wavelength λ	150
Figure 6-30: Normal solar absorptance, α_s , of Copper as a function of temperature T	151
Figure 6-31: Normal spectral absorptance, α'_λ , of Gold as a function of wavelength λ	152
Figure 6-32: Normal solar absorptance, α_s , of Gold as a function of temperature T	153
Figure 6-33: Normal spectral absorptance, α'_λ , of Molybdenum as a function of wavelength λ	154
Figure 6-34: Normal solar absorptance, α_s , of Molybdenum as a function of temperature T	155
Figure 6-35: Normal spectral absorptance, α'_λ , of Nickel as a function of wavelength λ	156
Figure 6-36: Normal solar absorptance, α_s , of Nickel as a function of temperature, T	158
Figure 6-37: Normal spectral absorptance, α'_λ , of Incoel as a function of wavelength λ	159
Figure 6-38: Normal solar absorptance, α_s , of Incoel as a function of the temperature, T , to which samples had been previously heated.	160
Figure 6-39: Normal spectral absorptance, α'_λ , of Platinum as a function of wavelength λ	161
Figure 6-40: Normal solar absorptance, α_s , of Platinum as a function of temperature, T	162
Figure 6-41: Normal spectral absorptance, α'_λ , of Silver as a function of wavelength, λ	163
Figure 6-42: Normal solar absorptance, α_s , of Silver as a function of temperature, T	164

Figure 6-43: Linear thermal expansion, $\Delta L/L$, of two nominally identical specimens of 6,35x10 ⁻⁶ m thick Mylar Double-Goldized as a function of temperature, T	172
Figure 6-44: Linear thermal expansion, $\Delta L/L$ of 6,35x10 ⁻⁶ - 7,62x10 ⁻⁶ m thick Kapton Double-Goldized, with Dacron Flocking, as a function of temperature, T	173
Figure 6-45: Coating thickness, t_c , given by several methods, compared with that gives by the electrical resistance method, $t_{c\Omega}$	181
Figure 6-46: Thickness, $t_{c\Omega}$, of metallic coatings as a function of film electrical resistance, R . Calculated by the compiler.....	183
Figure 6-47: Apparent emittance, ε_a , of a gray V-Groove as a function of surface emittance, ε , illustrating the effect of embossing or crinkling on the optical properties of the shield. Calculated by the compiler.	186
Figure 6-48: Effective thermal conductivity, k_{eff} , of several fibrous spacers as a function of mean temperature, T	199
Figure 6-49: Effective thermal conductivity, k_{eff} , of Fiber-glass batting as a function of Nitrogen gas pressure, p	200
Figure 6-50: Effective thermal conductivity, k_{eff} , of Dexiglas as a function of warm-boundary temperature, T_H	201
Figure 6-51: Effective thermal conductivity, k_{eff} , of Tissuglas as a function of warm-boundary temperature, T_H	202
Figure 6-52: Effective thermal conductivity, k_{eff} , of Refrasil as a function of warm-boundary temperature, T_H	203
Figure 6-53: Effective thermal conductivity, k_{eff} , of several spacer materials as a function of bulk density, ρ	204
Figure 6-54: Specific Heat, c , of several spacer materials as a function of temperature, T	205
Figure 6-55: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	211
Figure 6-56: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	212
Figure 6-57: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	215
Figure 6-58: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	216
Figure 6-59: Heat flux, Q/A , across a single-aluminized Mylar, crinkled, MLI on different substrate plates, as a function of the number of radiation shields, N . $T = 195$ K, $p = 2 \times 10^{-3}$ Pa. $N/t = 1100 \pm 200$ m ⁻¹	217
Figure 6-60: Effective thermal conductivity, k_{eff} , of a single-aluminized Mylar, crinkled, MLI on two different substrate plates, as a function of the number of radiation shields, N	218
Figure 6-61: Heat flux, Q/A , across a single-aluminized Mylar, crinkled, MLI on different substrate plates, as a function of the pressure, p . $T = 195$ K. Layer density, $N/t = 1100 \pm 200$ m ⁻¹ in any case.....	219

Figure 6-62: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	220
Figure 6-63: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	221
Figure 6-64: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	222
Figure 6-65: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	223
Figure 6-66: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. number of radiation shields per unit thickness, N/t	224
Figure 6-67: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	225
Figure 6-68: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	225
Figure 6-69: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t . Arrows in curves indicate whether the system is being loaded or unloaded.	228
Figure 6-70: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	229
Figure 6-71: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	229
Figure 6-72: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	231
Figure 6-73: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	232
Figure 6-74: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	234
Figure 6-75: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t . Complete loading-unloading history of system (Δ).	235
Figure 6-76: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	236
Figure 6-77: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t . Complete loading-unloading history of system (Δ).	236
Figure 6-78: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	239
Figure 6-79: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	240

Figure 6-80: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	243
Figure 6-81: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	244
Figure 6-82: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	245
Figure 6-83: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	246
Figure 6-84: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	247
Figure 6-85: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	248
Figure 6-86: Effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff} , vs. the number of radiation shields per unit thickness, N/t	249
Figure 6-87: Compressive mechanical load, P , on the multilayer insulation vs. the number of radiation shields per unit thickness, N/t	250
Figure 6-88: Effective thermal conductivity, k_{eff} , as a function of the characteristic temperature, T	250
Figure 6-89: Lateral effective thermal conductivity, k_{eff} , as a function of characteristic temperature, T	252
Figure 6-90: Lateral effective thermal conductivity, k_{eff} , as a function of characteristic temperature, T	253
Figure 6-91: Lateral effective thermal conductivity, k_{eff} , as a function of characteristic temperature, T	255
Figure 6-92: Lateral effective thermal conductivity, k_{eff} , as a function of characteristic temperature, T	256
Figure 6-93: Effective emittance, $\Delta \varepsilon_{eff}$, vs. overlap, I	267
Figure 6-94: Length, L^* , vs. overlap, I , of alternate layers.	267
Figure 6-95: Length, L^* , vs. underlap, I	268
Figure 6-96: Length, L^* , vs. k_{eff} . Edge rejection.....	268
Figure 6-97: Heat loss, ΔQ , due to stitching vs. the length of Stitch, L	270
Figure 6-98: Heat loss, ΔQ , due to stitch patching vs. the number of patch layers, N . Undisturbed system and test method as in Figure 6-97.....	271
Figure 6-99: Effect of perforations on the effective thermal conductivity of a multilayer insulation formed by 24 Radiation Shields $6,35 \times 10^{-6}$ m thick Mylar Double-Aluminized, and 23 Spacers $7,11 \times 10^{-4}$ m thick Polyurethane Foam.	272
Figure 6-100: Effect of perforations on the effective thermal conductivity of a multilayer insulation formed by 10 Radiation Shields $6,35 \times 10^{-6}$ m thick by 0,279 m diameter Mylar Double-Aluminized, and 22 Spacers $2,54 \times 10^{-5}$ m thick by 0,305 m diameter Glass Fabric.	273

Figure 6-101: Effect of percentage of perforations, τ , on the heat flux through a multilayer insulation.....	274
Figure 6-102: Effect of Meteoroid-Bumper debris damage on the effective thermal conductivity, k_{eff} , and product of apparent density and effective thermal conductivity, ρk_{eff}	276
Figure 6-103: Effective thermal conductivity, k_{eff} , as a function of gas pressure, p	278
Figure 6-104: Effective thermal conductivity, k_{eff} , as a function of gas pressure, p	280
Figure 6-105: Effective thermal conductivity, k_{eff} , as a function of gas pressure, p	281
Figure 6-106: Effective thermal conductivity, k_{eff} , as a function of gas pressure, p	283
Figure 6-107: Effective thermal conductivity, k_{eff} , as a function of gas pressure, p	284
Figure 6-108: Permeability, κ , of several multilayer insulation configurations as a function of the layer density, N/t	287
Figure 6-109: Pressure differential, $p(0,t)-p_o(t)$. Vs. time, t	290
Figure 6-110: Knudsen Diffusion Coefficient, D_K , as a function of gas molecular mass, M . From Coston (1967) [13], p. 4,3-31.	292
Figure 6-111: Outgassing rate, Q , as a function of pumping time, t , for Mylar Double-Aluminized. Effect of preconditioning. From Glassford (1970) [24].	293
Figure 6-112: Outgassing rate, Q , as a function of pumping time, t , for Mylar Double-Aluminized. Effect of prepumping. From Glassford (1970) [24].....	294
Figure 6-113: Outgassing rate, Q , as a function of pumping time, t , for Mylar Double-Aluminized, as received. Effect of test temperature.	295
Figure 6-114: Outgassing rate, Q , as a function of pumping time, t , for several shielding materials, as received.	297
Figure 6-115: Outgassing rate, Q , as a function of pumping time, t , for several spacing materials.	298
Figure 6-116: Outgassing rate, Q , as a function of pumping time, t , for an MLI system and for its components. No preconditioning. From Glassford (1970) [24].	299
Figure 6-117: Main characteristics of the different self-evacuated MLIs and experimental methods used to obtain the results summarized in Table 6-25....	303
Figure 6-118: Model for a guarded two-dimensional MLI. (a) Geometry. (b) Boundary conditions for the calculations.	317
Figure 6-119: Ratio of lateral to total heat transfer rate, $Q_x/(Q_x+Q_y)$, in an anisotropic two-dimensional continuous medium subject to the boundary conditions indicated in Figure 6-118:.....	318

Tables

Table 4-1: Relevant Properties of Ceramic Foams	25
Table 4-2: Properties of Ceramic Foams	29
Table 4-3: Characterization of Foams Whose Properties Will Be Given Later	38
Table 4-4: Average Tensile Data of Polyurethane and Polystyrene Foams	39
Table 4-5: Average Tensile Data of Polyurethane and Polystyrene Foams at $T = 76$ K T: Transverse; L: Longitudinal	40

Table 4-6: Average Tensile Data of Polyurethane and Polystyrene Foams at $T = 195\text{ K}$ T: Transverse; L: Longitudinal	41
Table 4-7: Average Tensile Data of Polyurethane and Polystyrene Foams at $T = 300$ KT: Transverse; L: Longitudinal	42
Table 4-8: Average Compressive Data of Polyurethane and Polystyrene Foams	42
Table 4-9: Average Compressive Data of Polyurethane and Polystyrene Foams at $T =$ 76 K T: Transverse; L: Longitudinal	43
Table 4-10: Average Compressive Data of Polyurethane and Polystyrene Foams at T $= 195\text{ K}$ T: Transverse; L: Longitudinal	44
Table 4-11: Average Compressive Data of Polyurethane and Polystyrene Foams at $T =$ 300 K T: Transverse; L: Longitudinal	45
Table 4-12: Information Concerning Processing	60
Table 4-13: Information Concerning Processing of Fiberfil Products Recommendations for Molding	63
Table 5-1: Available Densities, ρ , and Thicknesses, t , of B & W Kaowool Blanket. X denotes available.	90
Table 5-2: Available Thicknesses, t , and Lengths, L , of Fiberfrax Lo-Con Blanket and Felt for densities of 64 kg.m^{-3} and 96 kg.m^{-3} and roll widths 0,3 m, 0,61 m, and 1,22 m.	91
Table 5-3: Thermal Conductivity, k , [$\text{W.m}^{-1}.\text{K}^{-1}$], of J-M Thermoflex Felt for different values of Density, ρ , [kg.m^{-3}], and Mean Temperature, T_m , [K].	92
Table 5-4: Available Densities, ρ , [kg.m^{-3}], and Thicknesses, $t \times 10^3$ [m], of J-M Thermoflex.	95
Table 5-5: Available Densities, ρ , and Thicknesses, t , of J-M Micro-Quartz	96
Table 5-6: Available Forms of J-M Min-K 1301	102
Table 5-7: Available Forms of J-M Min-K 2000	102
Table 5-8: Available Thicknesses of B&W Kaowool Paper	104
Table 5-9: Available Thicknesses of Fiberfrax 970 Paper	105
Table 5-10: Available Roll Sizes of Fiberfrax 970 Paper	105
Table 5-11: Available Thicknesses and Corresponding Mass/Area of Fiberfrax Paper H880; Hi-Fi 660, and 550.	105
Table 6-1: Properties of Metallic Foils	166
Table 6-2: Properties of Polymeric Films	168
Table 6-3: Properties of Several Marketed Polycarbonate Resins	169
Table 6-4: Tensile Strength of Mylar and Kapton H for Different. Values of the Temperature.	170
Table 6-5: Properties of Coated Plastic Films	170
Table 6-6: Flammability of Several Radiation Shields in Oxygen and in Air	174
Table 6-7: Outgassing Characteristics of Non-Metallized and Metallized Plastic Films	175
Table 6-8: Procurement Data	177
Table 6-9: ρ and $3\lambda\rho/8$ for Aluminium, Copper, Gold and Silver	182

Table 6-10: Absorption and Scattering Coefficients for Spacer Materials	188
Table 6-11: Properties of Staple Fibers Data is available for the following Staple Fibers	190
Table 6-12: Properties of Staple Fibers	191
Table 6-13: Properties of Staple Fibers	192
Table 6-14: Effect of Heat on Several Fibers	193
Table 6-15: Moisture Regain of Several Fibers.....	193
Table 6-16: Effect of Exposure Temperature, Pressure and Time on the Strength of Fabrics, $T = 450$ K.....	194
Table 6-17: Fiber Strength after 10 min Exposure to Temperature	195
Table 6-18: Effect of Cycling vs. Continuous Exposure to Temperature of Two Fibers in Woven Forms.....	196
Table 6-19: Thermal Properties of Fabrics Woven of Several Fibers	197
Table 6-20: Flammability per Vertical Flame Tests	206
Table 6-21: Flammability of Several Spacers in Oxygen and in Air.....	206
Table 6-22: Outgassing Characteristics of Several Spacers	207
Table 6-23: Thermal Effects of Joints	259
Table 6-24: Characteristics of Self-Pumping MLIs the performances of which are given in Table 6-25	302
Table 6-25: Characteristics of Several Self-Pumping Multilayer Insulations.....	304
Table 6-26: Summary of the Best and Worst Cases Reached by the Different Investigators.....	314

European Foreword

This document (CEN/CLC/TR 17603-31-07:2021) has been prepared by Technical Committee CEN/CLC/JTC 5 "Space", the secretariat of which is held by DIN.

It is highlighted that this technical report does not contain any requirement but only collection of data or descriptions and guidelines about how to organize and perform the work in support of EN 16603-31.

This Technical report (TR 17603-31-07:2021) originates from ECSS-E-HB-31-01 Part 7A.

Attention is drawn to the possibility that some of the elements of this document may be the subject of patent rights. CEN [and/or CENELEC] shall not be held responsible for identifying any or all such patent rights.

This document has been prepared under a mandate given to CEN by the European Commission and the European Free Trade Association.

This document has been developed to cover specifically space systems and has therefore precedence over any TR covering the same scope but with a wider domain of applicability (e.g.: aerospace).

1

Scope

There are 3 main categories of insulators used in spacecrafts:

1. foams: organic and inorganic;
2. fibrous insulations: for internal and external insulation and for high temperature environments
3. multilayer insulations (MLI): layers of radiation reflecting shields.

Properties, thermal behaviour and application areas of the insulation materials used in spacecrafts are detailed in this Part 7.

The Thermal design handbook is published in 16 Parts

TR 17603-31-01	Thermal design handbook – Part 1: View factors
TR 17603-31-02	Thermal design handbook – Part 2: Holes, Grooves and Cavities
TR 17603-31-03	Thermal design handbook – Part 3: Spacecraft Surface Temperature
TR 17603-31-04	Thermal design handbook – Part 4: Conductive Heat Transfer
TR 17603-31-05	Thermal design handbook – Part 5: Structural Materials: Metallic and Composite
TR 17603-31-06	Thermal design handbook – Part 6: Thermal Control Surfaces
TR 17603-31-07	Thermal design handbook – Part 7: Insulations
TR 17603-31-08	Thermal design handbook – Part 8: Heat Pipes
TR 17603-31-09	Thermal design handbook – Part 9: Radiators
TR 17603-31-10	Thermal design handbook – Part 10: Phase – Change Capacitors
TR 17603-31-11	Thermal design handbook – Part 11: Electrical Heating
TR 17603-31-12	Thermal design handbook – Part 12: Louvers
TR 17603-31-13	Thermal design handbook – Part 13: Fluid Loops
TR 17603-31-14	Thermal design handbook – Part 14: Cryogenic Cooling
TR 17603-31-15	Thermal design handbook – Part 15: Existing Satellites
TR 17603-31-16	Thermal design handbook – Part 16: Thermal Protection System

2 References

EN Reference	Reference in text	Title
EN 16601-00-01	ECSS-S-ST-00-01	ECSS System - Glossary of terms

All other references made to publications in this Part are listed, alphabetically, in the **Bibliography**.

Terms, definitions and symbols

3.1 Terms and definitions

For the purpose of this Standard, the terms and definitions given in ECSS-S-ST-00-01 apply.

3.2 Abbreviated terms

The following abbreviated terms are defined and used within this Standard.

F	flammable
NF	non-flammable
LS	linear shrinkage, percent
MLI	multilayer insulation
SE	self-extinguishing
TWL	total weight loss, percent
VCM	volatile condensable materials, percent by weight
gpd	grams per denier

3.3 Symbols

A	area, [m]
C	coefficient for the degradation due to a joint, [m]
D	dissipation factor
D_K	knudsen diffusion coefficient, [m.s ⁻¹]
E	modulus of elasticity, [Pa]
E_c	compressive modulus of elasticity, [Pa]
E_f	flexural modulus of elasticity, [Pa]

F	propagating tear strength in a foil, [N]
F_{HC}	View factor, fraction of the total radiation which arrives at <i>C</i> coming from <i>H</i>
K₀	constant dependent on the interaction between gas and porous medium, [m]
L	length, [m]
L*	equivalent length for the degradation because of a joint, [m]
M	gas molecular mass, [kg.mol ⁻¹]
N	Clause 5: factor relating the radiative contribution to the effective thermal conductivity of a foam to the warm boundary temperature Clause 7: number of shields in an MLI, number of patch layers in an MLI
N_A	Avogadro constant, $N_A = 6,02252 \times 10^{23} \text{ mol}^{-1}$
P	applied compressive load, [Pa]
Q	heat flow through the insulation, [W] outgassing rate, [Pa.m.s ⁻¹]
R	electrical resistance of a foil, [Ω per Square] boltzmann constant, $R = 1,38054 \times 10^{-23} \text{ J.K}^{-1}$
R(2π,β')	reflectance factor, irradiation over a complete hemisphere
S	steric factor accounting for the distortion of the heat flow path due to foam expansion
T	temperature, [K]
T_m	arithmetic mean temperature, [K]
V	gas volume flow rate per unit area of the insulation, [m.s ⁻¹]
a	absorption coefficient, [m ⁻¹]
c	specific heat, [J.kg ⁻¹ .K ⁻¹]
c_p	gas specific heat at constant pressure, [J.kg ⁻¹ .K ⁻¹]
$\bar{c}$	average molecular speed, [m.s ⁻¹], $\bar{c} = (8RT/\pi m)^{1/2}$
d	diameter, [m]

f	frequency, [Hz]
h	thermal conductance, [W.m ⁻² .K ⁻¹]
k	thermal conductivity, [W.m ⁻¹ .K ⁻¹]
k_g	gas thermal conductivity, [W.m ⁻¹ .K ⁻¹]
k_{rad}	radiative contribution to the effective thermal conductivity, [W.m ⁻¹ .K ⁻¹]
m	mass of one molecule, [kg]
m	mass flow rate per unit cross-sectional area, [kg.m ⁻² .s]
m_h,	mass flow rate per unit cross-sectional area of an isolated hole, [kg.m ⁻² .s]
n,	refractive index
n_s	number of space layers per shield
p	gas pressure, [Pa]
	Clause 7: fraction of the electrons suffering specular reflection at the surface of a metallic film
p_o	ambient pressure, [Pa]
s	scattering coefficient, [m ⁻¹]
t	insulation thickness, [m]
t'	spacer thickness, [m]
t_c	coating thickness, [m]
t_Ω	coating thickness as measured by the electrical resistance method, [m]
t_u	uncompressed insulation thickness, [m]
t	time, [s] or [min]
w	width, [m]
x	geometrical coordinate
ΔL/L	linear thermal expansion
α	Clause 6: sound absorption coefficient. Clause 7: absorptance, thermal diffusivity
α_s	normal solar absorptance

α'_{λ}	directional spectral absorptance
β	Clauses 5 and 7: linear thermal expansion coefficient Clause 7: angle between surface normal and direction of incident flux, [angular degrees]
β'	angle between surface normal and direction of reflected, emitted or transmitted flux, [angular degrees]
δ	strain
ε	hemispherical total emittance
ε_a	apparent emittance of a groove
ε_r	dielectric constant
ε'	directional total emittance
θ	groove angle, [angular degrees]
κ	permeability of a porous medium, [m ²]
λ	wavelength of the incident radiation, [m] mean free path, [m]
μ	dynamic viscosity, [Pa.s]
π	pressure differential, [Pa]
ρ	density, [kg.m ⁻³] Clauses 5 and 7: reflectance Clause 7: bulk electrical resistivity, [Ω .m]
σ	Clauses 5 and 7: Stefan-Boltzmann constant, $\sigma = 5,6697 \times 10^{-8} \text{ W.m}^{-2}.\text{K}^{-4}$ Clause 5: strength, either tensile, compressive or flexural, [Pa]
σ_0	yield strength, [Pa]
τ	Clause 5: shear strength, [Pa] Clause 7: fractional area of perforations or transmittance of a perforated shield
τ_0	shear yield strength, [Pa]

Subscripts

C	cold boundary
H	hor boundary
e	concerns MLI evacuation by edge-controlled pumping
eff	effective
eff(nor)	normal effective ((nor) is omitted except when normal and lateral heat transfer appear simultaneously)
eff(lat)	lateral effective (same comment as for (nor))
h	concerns MLI evacuation by hole-controlled pumping
i	concerns the ith Layer or the ith joint of an MLI
ult	concerns ultimate strength

Superscripts

j	concerns species <i>j</i>
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4

Foams

4.1 General

Foams are commonly used as insulating materials in building, storage vessels, sports equipment, spacecraft, and others. They can be also used for potting and encapsulation of shock-sensitive components and for packaging. Because they have many advantages for cryogenic and space applications, efforts have been made to develop suitable foams; establish their physical, chemical and structural properties, and integrate foams with thermal insulation systems.

Because of their cellular structure of discrete or interconnected voids, a large percentage of the foam volume is air, or whatever gas used as the expanding agent, thence foams are light-weight and very low thermal conductivity, that could be decreased further by evacuation. In most cryogenic applications the insulation efficiency is achieved by condensation of the gases contained in the cellular structure of the plastic material (cryopumping).

The main shortcomings of the foams are that they are not gas-tight, that their mechanical properties are relatively poor, and that thermal stresses induce cracks which require the use of constrictive wraps to maintain the insulation integrity.

According to their chemical composition foams may be classified in inorganic and organic.

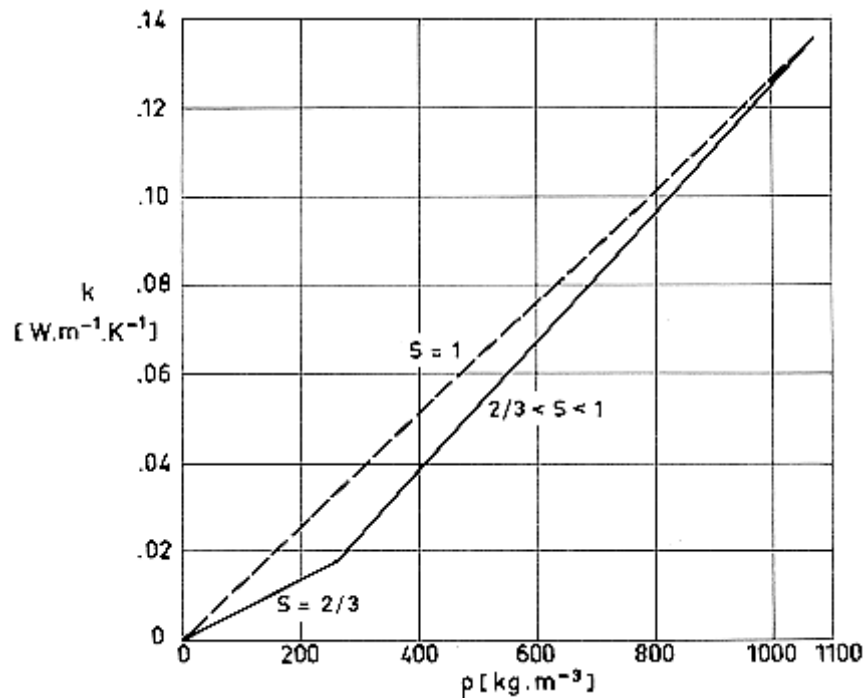
Inorganic foams have been used at elevated temperatures. These foams are difficult to produce and have inferior physical properties than organic foams at low densities (16 to 80 kg.m⁻³).

Organic foams, on the other hand, can be readily manufactured in low densities. These foams are used, in the range of cryogenic temperatures, either as a single insulation or combined with other materials to form an integrated system.

Three basic problems should be considered regarding the application of organic foams as thermal insulators, namely: thermal conductivity, thermal expansion and contraction, and flammability and explosion hazards.

1. Thermal Conductivity. Thermal conduction through a foam combines solid conduction, gas conduction and radiation. Although an effective thermal conductivity, in the Fourier law sense, can be measured with relative ease, analytical models are available for predicting the thermal conductivity and the net heat flow through candidate foam materials to be used as thermal insulators.

The solid thermal conductivity of most foams varies with temperature in an almost linear manner as with many other materials. When the density decreases, the solid conductivity will be reduced, not only because the decreasing of the solid cross section, but also by the influence of a steric factor since, even assuming that the shape of the cells is independent of the density, expansion elongates the paths through which heat flow. As an illustration Figure 4-1 shows the calculated variation of the thermal conductivity of a polyurethane resin as a function of foam density.

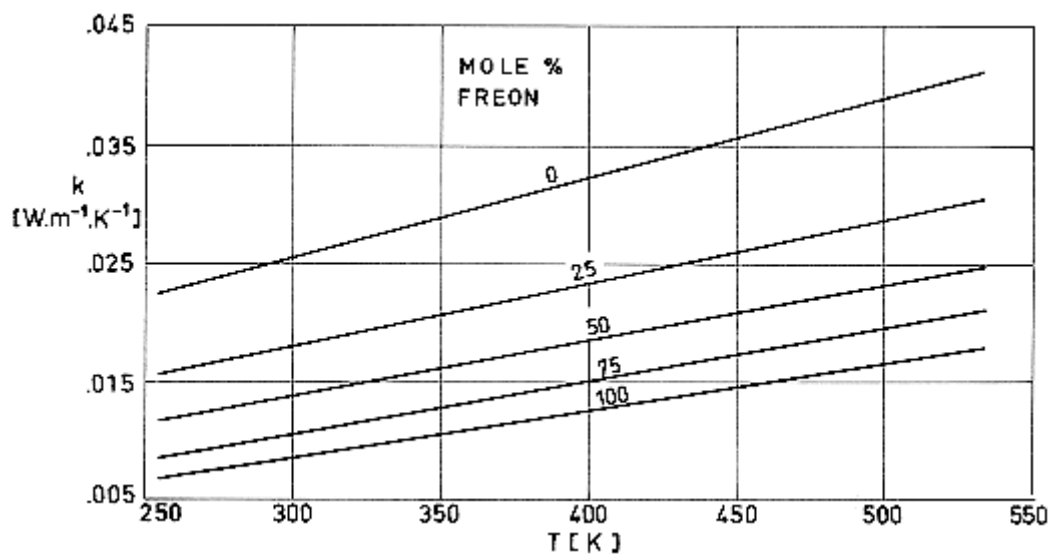


Note: non-si units are used in this figure

Figure 4-1: Resin thermal conductivity

k , as a function of foamed resin density, ρ . S is the steric factor which takes into account the distortion of the heat flow paths because expansion of the foam. Polyurethane resin. From Hammond (1969) [28].

The second mode of heat transfer of heat is heat conduction. In the case of closed-cell foams the gas thermal conductivity is a function of the concentration of both the foaming agent and the air which may diffuse into the foam outside. Since condensation can take place the thermal conductivities of both gases will depend not only on temperature but also on pressure. Figure 4-2 gives the gas thermal conductivity for mixtures of Freon 11, a typical foaming agent, and air, at atmospheric pressure, as function of temperature.



Note: non-si units are used in this figure

Figure 4-2 : Gas thermal conductivity

k , for mixtures of Freon 11 and air as a function of temperature, T , for different concentrations of Freon 11. From Hammond (1969) [28].

In most spacecraft applications the foam is outgassed, so that the gas conductivity is kept small enough to be neglected as compared with the other thermal conductivities. However, in the case of foams sprayed on cryogenic tanks for insulating purposes, the air has enough time to diffuse into the foam, possibly degrading the insulation characteristics in an amount which can be estimated. For example: in a typical case (Schroeder (1973)b [61]) a total cure of 72 hours is required, after spraying the foam, before the cryogenic exposure of the tank wall.

The radiative transfer of heat within the foam involves simultaneous transmission, absorption and scattering. According to Hammond (1969) [28] the radiative contribution to thermal conductivity of the foam can be written as

$$k_{rad} = N\sigma \frac{T_H^4 - T_C^4}{a(T_H - T_C)} = 4N\sigma \frac{T_H^3}{a} \quad [4-1]$$

where: N is a constant factor which lies between 0,5 (for low temperatures) and 2 (for high temperatures).

T_C , T_H are the cold and warm boundary temperatures, respectively. [K].

a is the foam absorption coefficient. [m^{-1}].

σ is the Stefan-Boltzmann Constant. [$W \cdot m^{-2} \cdot K^{-4}$].

Values of the radiation thermal conductivity are given in Figure 4-3 versus mean temperature, T_m . It can be deduced from this figure that radiation becomes significant above 350 K.

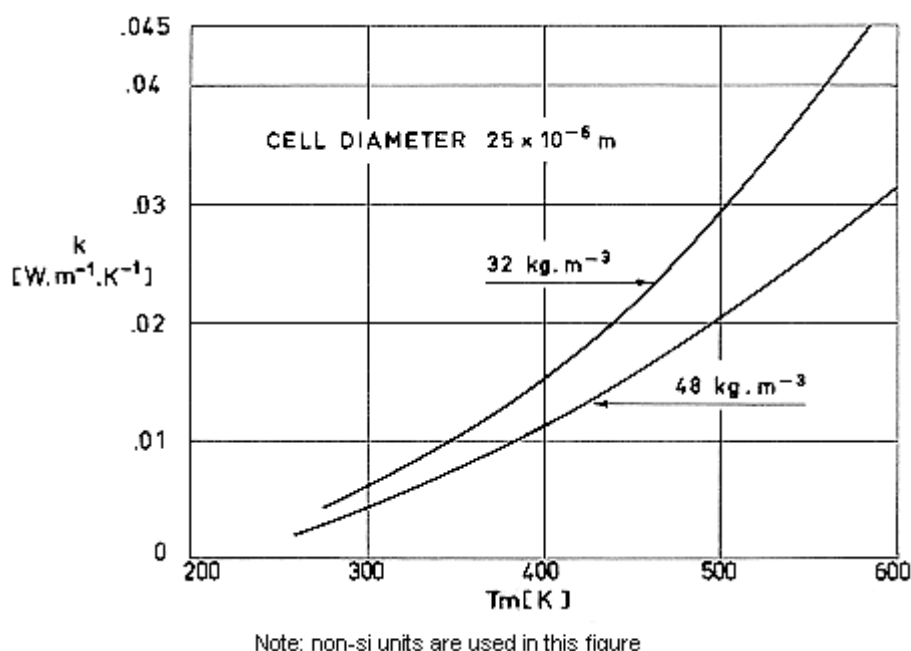


Figure 4-3: Radiation thermal conductivity

k_{rad} , as a function of mean temperature, T_m , for two polyurethane foams. From Hammond (1969) [28].

2. Thermal Expansion and Contraction. Broadly speaking, the thermal expansion coefficient of polyurethane foams is 2 to 5 times that of aluminium and 4 to 10 times that of steel. Because of this large difference in thermal expansions, the foam cannot be foamed in place against the metal or

bonded directly to it, since during chilldown from ambient temperature to cryogenic temperatures high stresses are created in the foam, and it fail either in shear along the tank surface or in tension in direction normal to it.

Various techniques for overcoming the effects of thermal contraction have been developed. For example, Glaser et al. (1967) [23], report that the addition of 10 percent (based on the weight of foam) of chopped glass fibers, $6,35 \times 10^{-3}$ to $12,7 \times 10^{-3}$ m long, to a low-viscosity one-shot foam eliminated cracking of the foam when exposed to liquid nitrogen. The same authors indicate that in work on the Saturn S-IV stage, where an internal insulation system was developed for the liquid-hydrogen propellant tank, urethane foam was reinforced with threads in order to prevent foam failure.

3. Fire and Explosion Hazards. The fire hazard of plastic materials has been demonstrated in the most dramatic fashion many times. Depending upon anticipated use, some polymeric materials are required to pass various flammability tests. The passing of these tests is assured by the addition of fire retardants.

When external foam insulation is used in spacecraft tankage, the possibility of ignition can be enhanced by the diffusion of air into foam. Because of the low temperatures the air condenses, and this oxygen rich condensed air, in conjunction with an oxidizable organic foam, can create an explosion hazard.

Flame-retardant coatings can be applied to the exterior of the insulation, once it has been cured, to enhance its self-extinguishing characteristics. When this precaution has been taken, the possibility of ignition or explosion in a well-designed foam insulation system is remote.

A further hazard associated with burning of foams is the toxicity of their combustion products, which can be ascribed to the fire-retardant additives. Petajan et al. (1975) [51] report that the physiological and toxicological effects of the combustion products of a laboratory-formulated fire-retarded rigid-polyurethane foam, proved fatal to rats in a matter of minutes, while the same rigid-polyurethane foam, without fire retardant, produced only minor signs of toxicity. The fire-retardant used in these experiments, a reactive phosphate, is not uncommon in the commercial formulations.

4.2 Inorganic foams

Porous ceramic foams have been developed to be used as high temperatures insulators, for space applications. These foams provide the high strength required in many instances. Nevertheless, their use is somewhat limited because of their high density and relatively high thermal conductivity.

Table 4-1 gives the relevant properties of several ceramics foams.

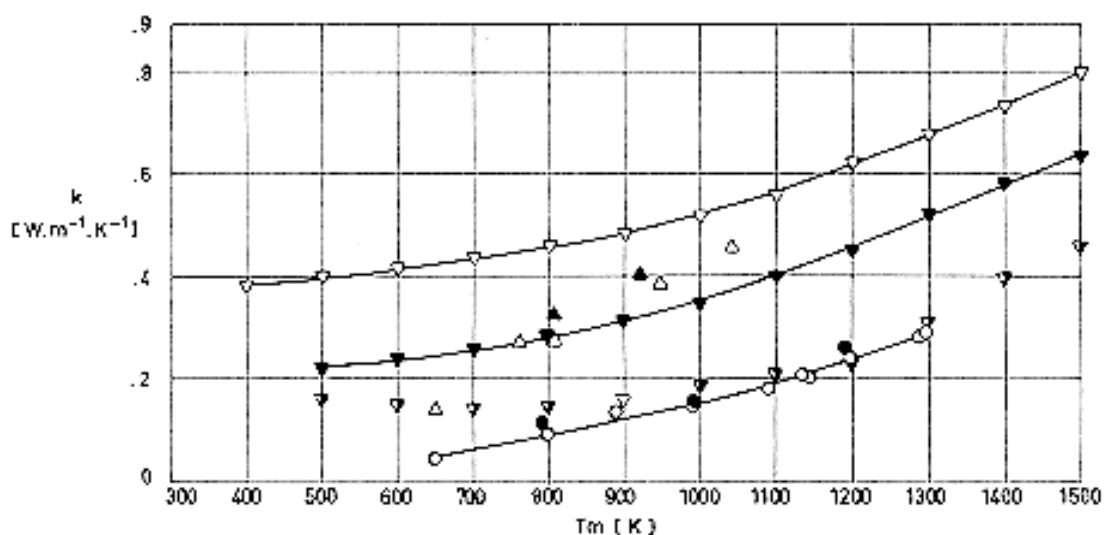
The influence of temperature on thermal conductivity and linear thermal expansion of the same types of foam is shown in Figure 4-4 and Figure 4-5 respectively.

Table 4-1: Relevant Properties of Ceramic Foams

	Alumina	Silica	Silicon Carbide	Zircon
Maximum Recommended Service Temperature [K]	2090	1920	2200	2470
Porosity [percent]	88	84	90	86
Density, $\rho \times 10^{-3}$ [kg.m ⁻³]	0,51	0,32	0,32	0,74

Thermal Conductivity, k , [W.m ⁻¹ .K ⁻¹]	0,605	0,158	0,821	0,137
Compressive Strength, $\sigma_{ult} \times 10^{-6}$ [Pa]	6,41	5,31	1,38	1,65

NOTE From Glaser et al. (1967) [23].



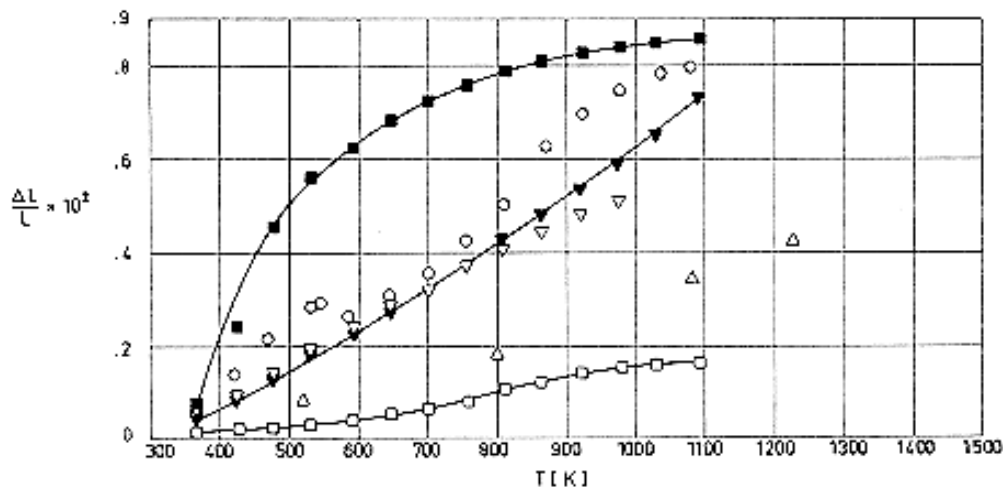
Note: non-si units are used in this figure

Figure 4-4: Thermal conductivity, k , of several ceramic foams as a function of arithmetic mean temperature, T_m .

Explanation

Key	Description	Comments	References
○	Alumina. Density, $\rho = 470 \text{ kg.m}^{-3}$.	Measured in $5,33 \times 10^{-5}$ to $2,66 \times 10^{-2}$ Pa pressure range.	Touloukian (1967)a [72].
●	Same as above.	Measured when cooled.	
△	Silicon Carbide. Density, $\rho = 460 \text{ kg.m}^{-3}$.	Measured in $5,33 \times 10^{-5}$ to 4×10^{-2} Pa pressure range.	Touloukian (1967)b [73].
▲	Same as above.	Measured when cooled.	
▽	Zircon. Density, $\rho = 960 \text{ kg.m}^{-3}$.	Data extracted from smooth curve.	Glaser et al. (1967) [23].
▼	Zircon Z-90. Density, $\rho = 940 \text{ kg.m}^{-3}$.		

▼	Zircon Z-67. Density, $\rho = 830 \text{ kg.m}^{-3}$.	
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Note: non-si units are used in this figure

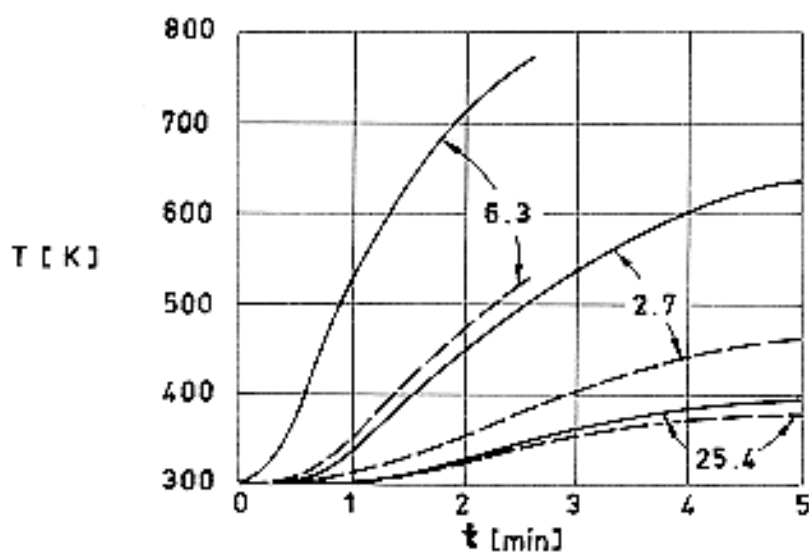
Figure 4-5: Linear thermal expansion, $\Delta L/L$, of several ceramic foams as a function of temperature, T .

Explanation

Key	Description	Comments	References
○	Alumina. 35-40 alumina hidrate, 25-30 Al_2O_3 , 22-27 H_3PO_4 (85%), 8-12 H_2O , 1-1,5 carbon powder, and 0,05-0,1 Al powder. Density, $\rho = 930 \text{ kg.m}^{-3}$. Dimensions, $5,08 \times 10^{-2} \text{ m}$ long by $6,35 \times 10^{-3} \text{ m}$ square.	Cured at 366 K; temperature raised continuously to 588,5 K during a 20 h period and then held at this temperature for 12 h. Reference temperature not given, assumed to be 293 K.	Touloukian (1967)a [72].
□	Silica. 50-55 Glasrock Slip, 30-35 Ludox (colloidal SiO_2), 5-7 SiO_2 400 mesh, 3-6 Al powder, 3-6 carbon powder, and 2-3 microballons. Density, $\rho = 640 \text{ kg.m}^{-3}$. Dimensions, $5,08 \times 10^{-2} \text{ m}$ long by $6,35 \times 10^{-3} \text{ m}$ square.	Cured at 366 K; temperature raised continuously to 588,5 K during a 24 h period and then held at this temperature for 8 h. Reference temperature not given, assumed to be 293 K.	
■	Same as above.	Second run.	
△	Silicon Carbide. Carborundum Company.	Measured in vacuum of $1,7 \times 10^3 \text{ Pa}$ up to 1200 K and with a slight	Touloukian (1967)b [73].

	Density, $\rho = 610 \text{ kg.m}^{-3}$. Dimensions, $7,62 \times 10^{-2} \text{ m}$ long by $1,27 \times 10^{-2} \text{ m}$ diameter.	purge of argon above 1200 K.	
▽	Zircon. 30-35 ZrO ₂ 100 mesh, 20-25 ZrO ₂ 325 mesh, 20-25 of 90 rare-earth oxide, 10- 12 H ₃ PO ₄ , 10-12 H ₂ O, 1-2 carbon powder, and 0,05-0,1 Al powder. Density, $\rho = 1600 \text{ kg.m}^{-3}$. Dimensions, $5,08 \times 10^{-2} \text{ m}$ long by $6,35 \times 10^{-3} \text{ m}$ square.	Cured at 366 K; temperature raised continuously to 588,5 K during a 20 h period and then held at this temperature for 24 h. Reference temperature not given, assumed to be 293 K.	Touloukian (1967)c[74] .
▼	Same as above.	Second run.	

Zircon foams that can be cast and later cured in place at temperatures below 390 K have been developed. These foams have excellent insulating and structural properties and are suitable for use in rigid heat shields. Figure 4-6 shows the temperature evolution of the hot face, T_H , and back face, T_C , of several pieces of zircon foam which are heated by a heat flux of $4,5 \times 10^5 \text{ W.m}^{-2}$. To perform the tests the foams were placed in contact with a stainless-steel honeycomb panel exposed to the heat flux.



Note: non-si units are used in this figure

Figure 4-6: Temperature evolution of the hot and cold faces of several pieces of Zircon foam. Solid line: T_H , hot face. Dashed line: T_C , cold face.

The numbers in the figure indicate the foam thickness in mm. From Glaser et al (1967) [23].

Impregnated porous ceramics may also be used as insulating materials (Glaser et al, (1967) [23]). These foams consist of a refractory porous base uniformly impregnated with a solid substance which sublimates or decomposes at elevated temperatures. Phenolics, and thermoplastics such as nylon, polyethylene, and polypropylene, have been used as the impregnating material. The composite is very similar to ablative substances used in reentry heat shields; although it presents the advantage that no changes in the shape of the surface occur.

Thermal switches formed by ceramic foams impregnated with metals such as silver or copper can be used to protect radioisotope capsules from reentry heating (Bustard, Princiotta & Barr (1970) [9]). These composites are thermal conductors ($k \cong 9 \text{ W.m}^{-1}\text{.K}^{-1}$) below the melting point of the metal impregnate and switch to insulators ($k \cong 0,2 \text{ W.m}^{-1}\text{.K}^{-1}$) above its melting point. In this manner, the thermal conductance during normal power system operation is relatively high, facilitating the transfer of heat from the fuel capsule to the hot side of the energy conversion device, without excessive capsule and/or radioisotope temperatures. During reentry into the earth's atmosphere the capsule is protected against the aero thermodynamic heating by the unimpregnated ceramic foam.

Table 4-2, which has been borrowed from Bustard et al. (1970) [9], gives additional properties of several ceramic foams.

Table 4-2: Properties of Ceramic Foams

Description	Alumina Ipsen 3300	Alumina Ipsen 3400	Alumina Astro Met 200	Alumina Insalute	Fused Silica Glassrock 25
Chemical Formula	Al ₂ O ₃	Al ₂ O ₃	Al ₂ O ₃	Al ₂ O ₃	SiO ₂
Thermal Conductivity, k , at 1350 K, [W.m ⁻¹ .K ⁻¹]	0,64	0,52	0,69	0,40	0,28
Specific Heat, $c \times 10^{-3}$ at 1350 K, [J.kg ⁻¹ .K ⁻¹]	1,34	1,34	1,34	1,34	1,46
Melting Point [K]	2320	2320	2320	2320	2000
Density, $\rho \times 10^{-3}$ [kg.m ⁻³]	0,92	0,56	0,39	0,85	0,42
Pore Structure	Coarse pore-bubble structure	Coarse pore-bubble structure	Open pore-lacey structure	Irregular pore	Fine pore-bubble structure
Thermal Shock Resistance	Good-bairline cracks developed	Fair to poor-spalling	Very poor-severe cracking	Fair to poor-cracking and spalling	Excellent-no structural damage
Compressive Strength, $\sigma_{ult} \times 10^{-6}$ [Pa]	6,20	4,83	0,34	2,31	3,10
Shear Strength, $\tau_{ult} \times 10^{-6}$ [Pa]	3,59	3,03	0,69	1,24	1,72

Machinability	Fair	Good	Very poor	Fair	Excellent
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Description	Fused Silica Thermo Matls. 50	Fused Silica Thermo Matls. HP50	Zircon Ipsen 4400	Zircon Ipsen 4300
Chemical Formula	SiO ₂	SiO ₂	ZrO ₂ ^a	ZrO ₂ ^b
Thermal Conductivity, <i>k</i> , at 1350 K, [W.m ⁻¹ .K ⁻¹]	0,38	0,38	0,40	0,40
Specific Heat, <i>c</i> x10 ⁻³ at 1350 K, [J.kg ⁻¹ .K ⁻¹]	1,46	1,46	0,75	0,75
Melting Point [K]	2000	2000	2960	2960
Density, ρ x10 ⁻³ [kg.m ⁻³]	0,80	0,80	1,45	1,10
Pore Structure	Fine pore-bubble structure	Fine pore- bubble structure	Coarse pore- bubble structure	Coarse pore- bubble structure
Thermal Shock Resistance	Excellent-no structural damage	Excellent-no structural damage	Good-no structural damage	Poor-severe cracking
Compressive Strength, σ_{ult} x10 ⁻⁶ [Pa]	12,1	22,4	6,14	5,51
Shear Strength, τ_{ult} x10 ⁻⁶ [Pa]	3,45	5,38	3,72	3,79
Machinability	Excellent	Excellent	Fair	Good

^a Partially stabilized to cubic structure.

^b Fully stabilized to cubic structure.

NOTE From Bustard, Princiotta & Barr (1970) [9].

4.3 Organic foams

Low density organic foams are used for cryogenic insulation because they offer the following advantages:

1. Ease of fabrication,
2. Relatively low cost, and
3. Self-supporting structure.

Like plastics, organic polymer foams can be divided into two basic groups: thermoplastic foams and thermosetting foams.

Thermoplastic foams soften at temperatures of 340 to 390 K and, in most cases, tend to become brittle at cryogenic temperatures. These foams can be made from such polymers as cellulose acetate, polystyrene, styrene-acrylonitrile copolymer, polyvinylchloride, polyethylene, and polypropylene. Among these foams polystyrene and polyvinyl-chloride are the most commonly used. Polystyrene foams are rigid and polyvinyl chloride are flexible, although rigid PVC foams have been developed recently which are competitive with other rigid foams.

Thermosetting foams are commonly made with phenolic, urea-formaldehyde, epoxy, and polyurethane resins. Flexible polyurethane foams dominate the market; these foams exhibit very good properties at cryogenic temperatures with a minimum tendency to become brittle. Urea-formaldehyde foams are brittle and do not show good structural properties at low densities. Low density phenolic resin foams tend to be brittle and friable at room temperature and are difficult to manufacture. Epoxy foams have met only with limited use because of difficulties in controlling satisfactorily the foaming.

Tetrafluoroethylene foam (TFE) is an intermediate material between a thermosetting and a thermoplastic foam. It melts at very high temperature, has very good physical properties at cryogenic temperatures and, because of its extreme chemical inertness, resists the effects of liquid oxygen, hydrazine, and other chemically reactive rocket fuels.

4.3.2 Thermal properties of organic foams

Figure 4-7 summarizes the available information on the thermal conductivity of polyurethane foams in a wide range of temperatures, and for different foam densities and preconditioning. Data for polystyrene foams are given in Figure 4-8.

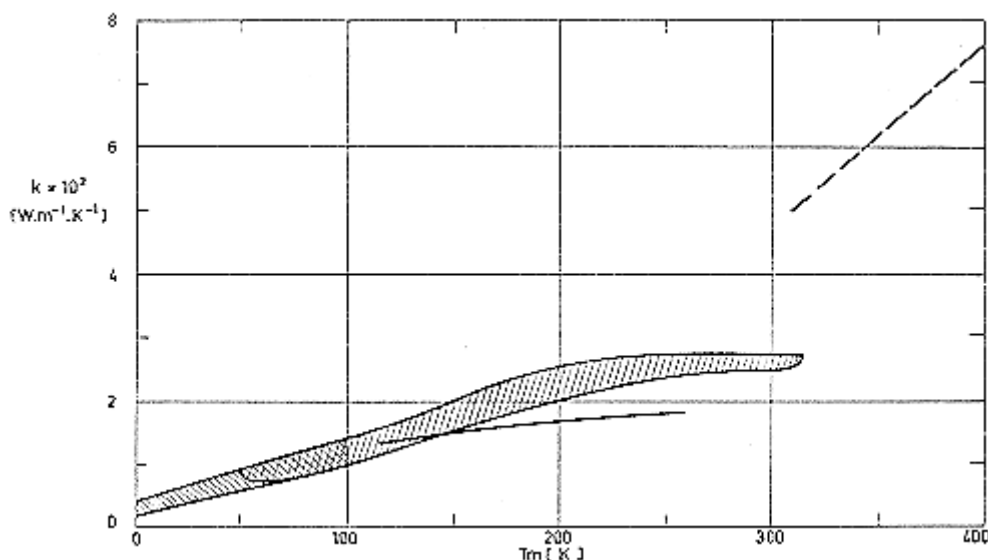


Figure 4-7: Thermal conductivity k , of polyurethane foams vs. arithmetic mean temperature, T_m .

Left-oriented shadowed area: $64 \text{ kg.m}^{-3} > \rho > 32 \text{ kg.m}^{-3}$; cryopumped; from Coston (1967).

Right-oriented shadowed area: From Rosato (1968) [56].

Solid line: CO_2 blown, $\rho = 32 \text{ kg.m}^{-3}$; evacuated; from Schroeder (1973)b [61].

Dashed line: The same foam at ambient pressure in air; from Schroeder (1973)b [61].

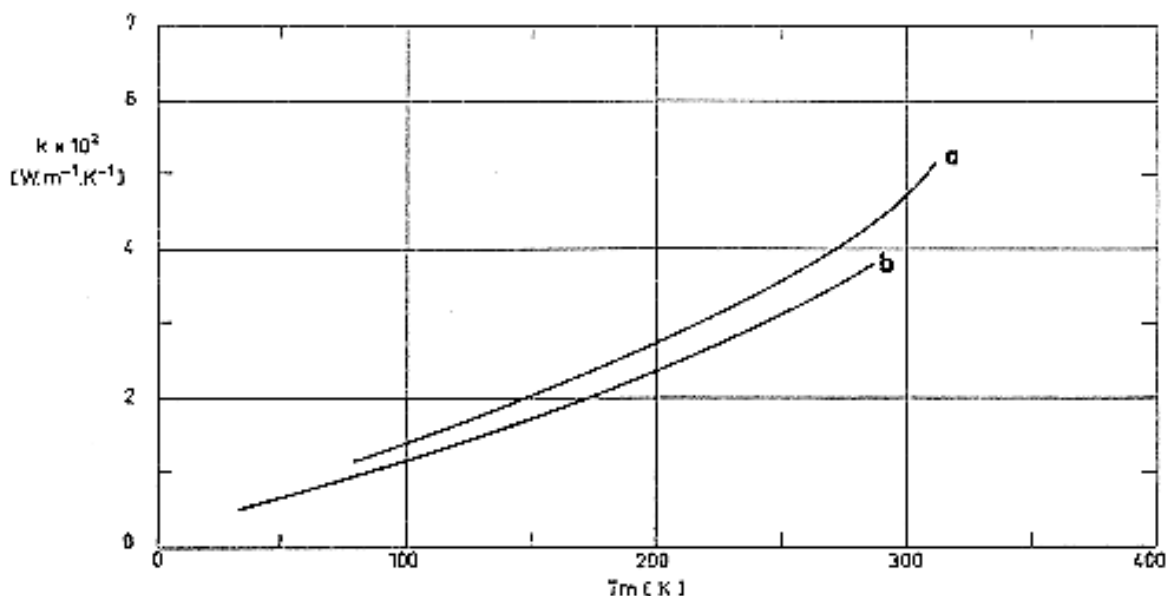


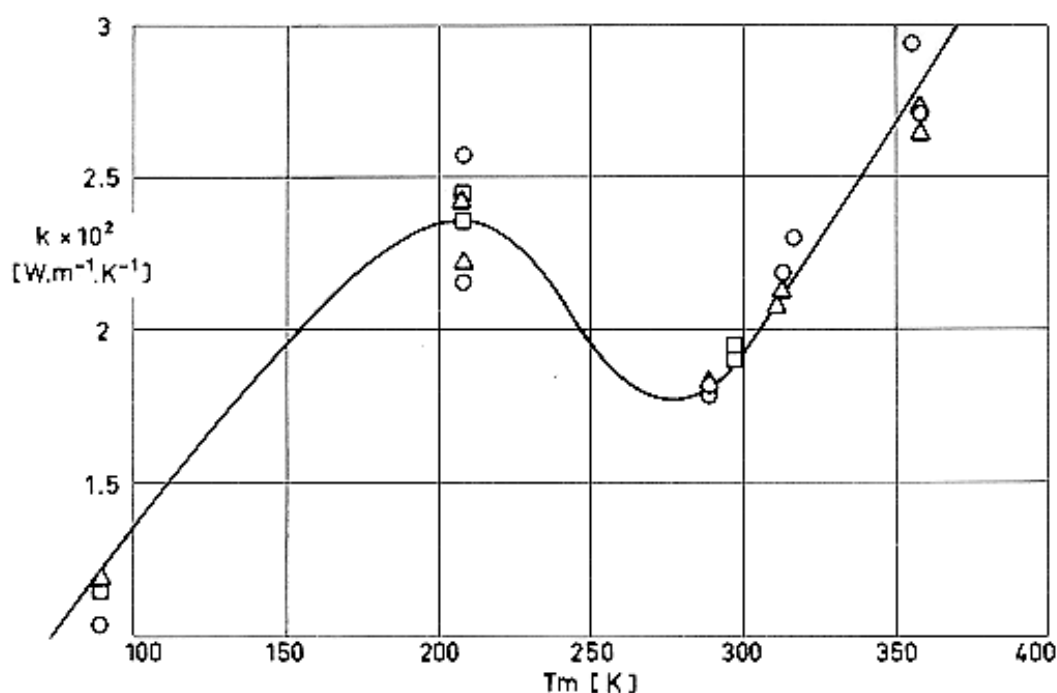
Figure 4-8: Thermal conductivity, k , of cryopumped polystyrene foams.

a: Styrofoam, $\rho = 33 \text{ kg.m}^{-3}$. *b*: Styrofoam, $\rho = 22 \text{ kg.m}^{-3}$. Styrofoam is a registered trade name of Dow Chemical Co. From Rosato (1968) [56].

As has been already said, the thermal conductivity of foams can be reduced further by evacuation.

Evacuation by vacuum pumps is used when foams are employed as spacers of multilayer insulations. However, for insulating relatively large cryogenic tanks pump evacuation is normally impractical, nevertheless, in that case the gas contained in the cellular structure condenses because of the prevailing low temperature level, thus leaving the void space evacuated (cryopumping).

A close examination of the thermal conductivity-temperature curve in the proximity of the condensation temperature of the filling gas reveals the typical S-curve behavior shown in Figure 4-9. This particular curve corresponds to a Freon 11-air filled foam. At atmospheric pressure Freon 11 condenses between 260 and 290 K. At temperatures below 260 K air controls the gas conduction contribution to the foam thermal conductivity. This contribution decreases when the temperature decreases, reaching very low levels when the air condenses. In the temperature range above the Freon condensing temperature the gas conduction depends on the concentration of Freon in the binary Freon-air system, and on temperature, increasing when the temperature increases, as occurs with single-phase gaseous systems. In the intermediate temperature range, when Freon starts boiling, the conductivity of the gaseous system decreases since the thermal conductivity of Freon is about a fourth of that of air.



Note: non-si units are used in this figure

Figure 4-9: Thermal conductivity, k , vs. arithmetic mean temperature, T_m , of a polyurethane foam in the proximity of the condensation temperature of the filling gas.

The temperature gradient is parallel to the foam rise direction. Foam: Nopcofoam BX-250, two-component, flame retardant, spray foam. Filling gas: Freon 11.

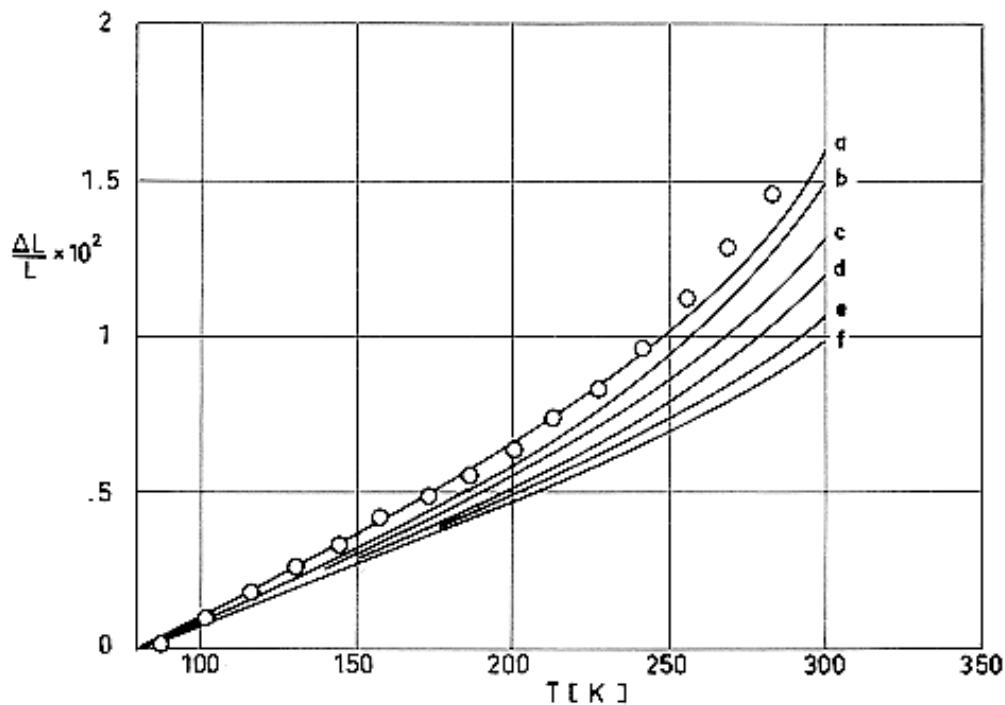
○: $\rho = 30,6 \text{ kg.m}^{-3}$.

□: $\rho = 32,7 \text{ kg.m}^{-3}$.

△: $\rho = 37,5 \text{ kg.m}^{-3}$.

From Schroeder (1973)b [61].

Linear thermal expansion of relevant foams is shown in Figure 4-10. It can be seen that Epoxy foams have the lowest thermal expansion. For a given foam, the thermal expansion increases when the temperature increases. Especially significant is the fact that in the cryogenic range all the foams have thermal expansion characteristics which can be reasonably considered as linear functions of temperature, the nonlinearities becoming apparent at room temperature.



Note: non-si units are used in this figure

Figure 4-10: Linear thermal expansion, $\Delta L/L$, of several organic foams as a function of temperature, T .

○: Polyurethane foam. $\rho = 50 \text{ kg.m}^{-3}$. From Schroeder (1973)b [61].

a Urethane foam. $\rho = 80 \text{ kg.m}^{-3}$.

b Urethane joint filler.

c Syntactic foam. Urethane filler with hollow silica spheres.

d Semiflexible Urethane.

e Epoxy.

f Epoxy.

Curves *a* to *f* are from Glasser et al. (1967) [23].

4.3.3 Mechanical properties of organic foams

In general, the main function of an insulation system is to provide thermal protection; it is not intended to be a primary load-carrying structure. However, since the insulation system is normally bonded or mechanically attached to the primary structure, it should match the strains and deflections of that structure, and therefore significant stresses could be introduced in it.

Low temperature mechanical property data of foams are extremely scarce, notwithstanding the fact that they are required for efficient design of foam-based insulations. Because of these reasons it seems advisable to collect these data in the present item.

The tensile strengths of a number of foams at temperatures ranging from 400 K down to 20 K are shown in Figure 4-11. At 20 K both the semirigid polyurethane and the epoxy foams have tensile strengths of the order of $4 \times 10^5 \text{ Pa}$. At room temperature the rigid polyurethane foam has the highest tensile strength, closely followed by the semirigid polyurethane foam. At room temperatures of the order of 400 K the tensile strength of all the organic foams practically vanishes.

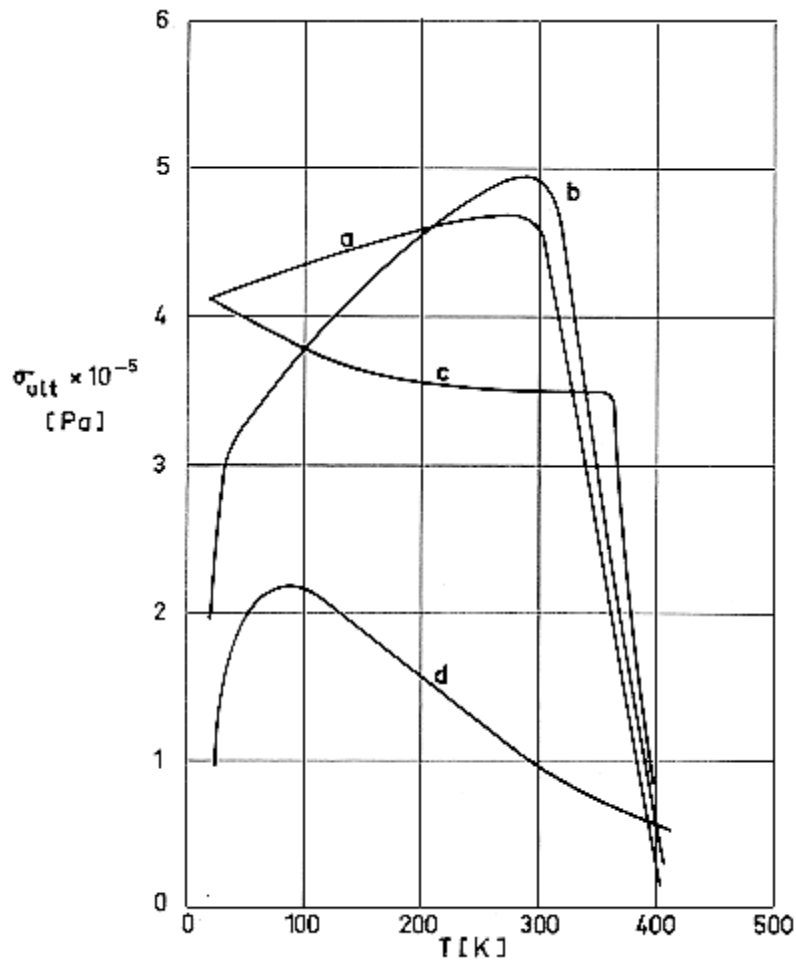


Figure 4-11: Ultimate tensile strength, of several foams as a function of temperature, T .

a: Barret semirigid polyurethane.

b: Hexcel 1414-2 polyester polyurethane rigid.

c: Magnolia 7015 epoxy.

d: Baron S-39 flexible polyester.

From Rosato (1968) [56].

The results of shear strength tests are summarized in Figure 4-12. All the foams tested exhibit minimum shear strengths at the two extreme temperatures. Again the rigid polyurethane foam has the highest shear strength among all of them.

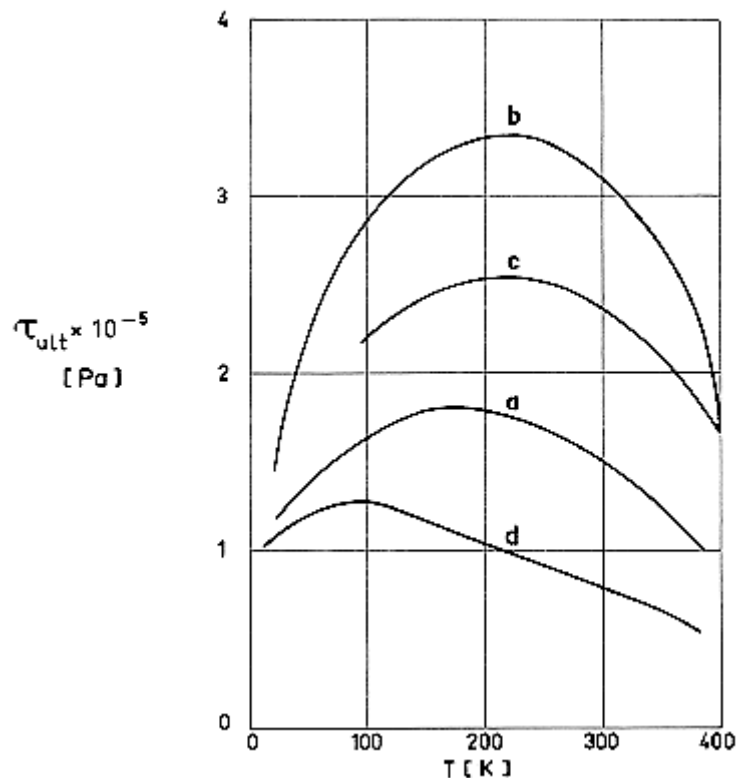


Figure 4-12: Ultimate shear strength, τ_{ult} , of several foams as a function of temperature, T .

- a*: Barret semirigid polyurethane.
b: Hexcel 1414-2 polyester polyurethane rigid.
c: Magnolia 7015 epoxy.
d: Baron S-39 flexible polyester.
 From Rosato (1968) [56].

The above or similar figures could give some order of magnitude idea useful for comparison purposes, however, the available information indicates that mechanical properties of foams are very sensitive to foam density, direction of cell-rise versus direction of loading, and test method.

All organic materials tested to date behave in a viscoelastic manner. Strictly speaking the thermal stresses caused on these materials by differential expansions or contractions cannot be expressed in terms of the standard modulus of elasticity and the free contraction constant.

Test methods have been devised to measure the thermal stress-thermal strain ratio (thermoelastic modulus) and the thermoelastic contraction as a function of temperature (Schroeder (1973) [60]). The main objection of such type of data is that because of the nonlinearity of the phenomenon it is not easy to superpose thermally induced and mechanically induced stresses. For example, tests reveal that the thermoelastic strain of a loaded specimen is not the same as that corresponding to the unloaded or free shrink specimen.

One way to circumvent these difficulties and still generate data which would be useful for engineering purposes, is to neglect non-linear effects, which at low temperatures seems to be justified, as was mentioned in connection with Figure 4-10 and is confirmed later, and to obtain mechanical property data using normal widely used test techniques, with the obvious additions to reproduce the required low temperature environment and to minimize thermally induced stresses.

A collection of data generated by US National Bureau of Standards is presented in Table 4-3 to Table 4-8 and Figure 4-13 to Figure 4-18 below. These data concern both tensile and compressive strengths. The main results of these tests can be summarized as follows:

1. The Modulus of Elasticity, yield strength, and ultimate strength increase with decreasing temperature, while the elongation decreases, Table 4-4 and Table 4-8 below.
2. The Modulus of Elasticity is twice as great in tension compared to compression, but the difference diminishes at lower temperatures. Compare Figure 4-14 and Figure 4-17 below.
3. Most foams become brittle at low temperatures, whereas the ductility is considerable at 195 K and above (Figure 4-13 and Figure 4-16 below). On the average, polystyrene foams are more ductile than polyurethane foams at 76 K (Figure 4-16 below).
4. The Modulus of Elasticity and the strength, both tensile and compressive, are approximately linearly dependent on density. At low temperatures the density dependence is greater (Figure 4-14, Figure 4-15, and Figure 4-17, Figure 4-18 below). In the compressive case there is a minimum density below which the rigid foam cannot withstand any applied compressive force at low temperatures; by extrapolation of the straight lines corresponding to circle and inverted triangle data points in Figure 4-17 below, one obtains a zero value of E at finite density.
5. Specimens loaded along the rise are stronger than those loaded transversally to rise (Figure 4-13 and Figure 4-16 below). In the compressive case this is not necessarily true; notice, for instance, the foam YR-1 in Figure 4-16 below.
6. At most densities there is considerable data spread which can be attributed to foam components, manufacturing process, and blowing agent.

Table 4-3: Characterization of Foams Whose Properties Will Be Given Later

	Sample a	Type b	Density, ρ [kg.m ⁻³]	Polyol Component	Isocyanate Component ^c	Blowing Agent	Flame Retardant
Flexible	AF-1	PU	34,8	Polyether	TDI	All CO ₂	No
	BF-4	PU	125	Polyether	PAPI/TDI		No
	BF-5	PU	123	Polyether	PAPI/TDI		No
	BF-6	PU	116	Polyether	PAPI/TDI		No
	CF-7	PU	33,8	Polyester	TDI		No
	CF-8	PU	32,7	Polyester	TDI		No
	CF-9	PU	33,5	Polyester	TDI		No
	DF-1	PU	110	Polyester	TDI		No
	DF-2	PU	111	Polyester	TDI		No
	DF-3	PU	105	Polyester	TDI		No
Rigid	JR-1	PU	66,2	Polyether	PAPI Prepol	CO ₂ and Freon 11	Yes
	KR-1	PU	30,6	Polyether	PAPI Prepol		Yes
	LR-3	PU	42,4	Polyether	TDI Prepol		No
	MR-1	PU	68,4	Polyether	TDI Prepol		No
	NR-1	PU	95,6	Polyether	TDI Prepol		No
	OR-1	PU	47,6	Polyester	TDI Prepol		No
	OR-2	PU	47,7	Polyester	TDI Prepol		No
	PR-1	PU	48,1	Polyether	PAPI		No

	Sample ^a	Type ^b	Density, ρ [kg.m ⁻³]	Polyol Component	Isocyanate Component ^c	Blowing Agent	Flame Retardant
	SR-2	PU	47,7	Polyether	PAPI	Freon 11	Yes
	TR-2	PU	32,4	Polyester	TDI Prepol	Freon 11	No
	UR-2	PU	43,6	Polyester	TDI Prepol	Freon 11	No
	VR-2	PU	61,0	Polyester	TDI Prepol	CO ₂	No
	WR-2	PU	87,8	Polyester	TDI Prepol	CO ₂	No
	XR-3	PU	48,2	Polyether	TDI Prepol	CO ₂	No
	XR-4	PU	47,6	Polyether	TDI Prepol	CO ₂	No
	YR-1	PS	52,2	Polystyrene	-	CO ₂	No
	ZR-1	PS	100	Polystyrene	-	CO ₂	No
						Pentane	
						Pentane	

^a Material source: AF-1 to SR-2: CPR Division, Upjohn Company; TR-2 to Zr-1: Bendix Corporation

^b PU: Polyurethane foam; PS: Polystyrene foam.

^c TDI: Tolyene diisocyanate; PAPI: Polymethylene polyphenylisocyanate, registered trade name of Upjohn Co.; Prepol: Prepolymerized.

NOTE From Reed, Arvidson & Durcholz (1972) [52].

Table 4-4: Average Tensile Data of Polyurethane and Polystyrene Foams

Data at $T = 76$ K	Table 4-5
Data at $T = 195$ K	Table 4-6
Data at $T = 300$ K	Table 4-7

NOTE From Reed, Arvidson & Durcholz (1972) [52].

**Table 4-5: Average Tensile Data of Polyurethane and Polystyrene Foams at $T = 76$
K T: Transverse; L: Longitudinal**

Foam	Density, ρ [kg.m ⁻³]	Type ^a	Modulus of Elasticity, $E \times 10^{-6}$ [Pa]		Proportional limit $\sigma \times 10^{-5}$ [Pa]		Yield ^b strength, $\sigma \times 10^{-5}$ [Pa]		Compressive strength, $\sigma \times 10^{-5}$ [Pa]		Elongation %			
											Plastic		Total	
			T	L	T	L	T	L	T	L	T	L	T	L
KR-1	30,6	PUR	12,5	-	3,2	-	-	-	3,7	-	0,06	-	3,05	-
KR-1	29,5	PUR	-	18,2	-	1,4	-	-	-	4,6	-	0,03	-	1,81
TR-2	32,4	PUR	20,4	-	3,0	-	-	-	3,6	-	0,03	-	1,80	-
LR-3	42,4	PUR	38,5	-	6,8	-	-	-	8,3	-	0,04	-	1,86	-
UR-2	43,6	PUR	33,4	-	4,7	-	-	-	6,0	-	0,01	-	1,88	-
UR-2	44,5	PUR	-	27,1	-	4,1	-	-	-	6,1	-	0,07	-	2,40
XR-4	47,6	PUR	31,7	-	-	-	-	-	5,6	-	0,00	-	1,77	-
OR-1	47,7	PUR	37,0	-	-	-	-	-	10,1	-	0,00	-	2,72	-
OR-1	46,3	PUR	-	45,0	-	-	-	-	-	6,3	-	0,00	-	2,03
SR-2	47,7	PUR	38,7	-	-	-	-	-	6,6	-	0,00	-	1,67	-
PR-1	47,7	PUR	32,8	-	5,0	-	-	-	5,4	-	0,02	-	1,70	-
VR-2	61,0	PUR	51,0	-	6,3	-	-	-	9,2	-	0,02	-	1,62	-
JR-1	64,6	PUR	40,4	-	-	-	-	-	7,9	-	0,00	-	1,93	-
MR-1	68,2	PUR	44,5	-	6,5	-	-	-	9,4	-	0,05	-	2,17	-
WR-2	87,8	PUR	63,0	-	11,0	-	-	-	14,4	-	0,05	-	2,22	-
NR-1	95,6	PUR	95,8	-	14,4	-	-	-	18,5	-	0,04	-	1,98	-
NR-1	94,8	PUR	-	85,4	-	13,2	-	-	-	15,7	-	0,06	-	2,21
CF-7	33,8	PUF	16,5	-	1,9	-	2,7	-	4,4	-	1,79	-	4,63	-
CF-7	32,5	PUF	-	17,6	-	2,6	-	3,7	-	4,9	-	0,83	-	3,54
AF-1	34,8	PUF	15,4	-	2,0	-	2,8	-	4,2	-	2,40	-	5,35	-
DF-1	108	PUF	102,7	-	6,6	-	12,3	-	26,3	-	3,27	-	5,90	-
BF-4	124	PUF	120,1	-	7,8	-	-	-	10,4	-	0,08	-	0,73	-
YR-1	52,2	PSR	57,4	-	7,2	-	9,9	-	11,2	-	0,72	-	2,02	-
YR-1	51,3	PSR	-	50,4	-	6,5	-	9,4	-	10,9	-	0,83	-	2,95
ZR-1	101	PSR	108,2	-	11,8	-	19,2	-	22,5	-	0,76	-	2,18	-

^a PUR: polyurethane rigid; PUF: polyurethane flexible; PSR: polystyrene rigid.

^b At 0,2% offset.

NOTE COMMENTS: 1) The rigid foam samples were cylindrical, $2,866 \times 10^{-2}$ m in diameter and $7,62 \times 10^{-2}$ m long. Specimens having a reduced middle subclause $6,21 \times 10^{-2}$ m in diameter and $2,54 \times 10^{-2}$ m long m in side and $7,62 \times 10^{-2}$ m long. The specimens were epoxied to polycarbonate grips. Polycarbonate was chosen because its thermal expansion matched that of the adhesive at low temperature.
2) The rate of head movement of the testing machine was 5×10^{-4} m.min⁻¹.

3) Each value represents the average of about four tests. Strength data variation from the reported average were $\pm 10\%$. Suggested inaccuracies of the reported average strength data are $\pm 5\%$. Average elongation data are supposed to be inaccurate to $\pm 5\%$, and data variations from the reported averages are $\pm 10\%$. For a given test individual strength and elongation measurements are thought to be accurate to $\pm 1\%$.

NOTE From Reed, Arvidson & Durcholz (1972) [52].

Table 4-6: Average Tensile Data of Polyurethane and Polystyrene Foams at $T = 195$
K T: Transverse; L: Longitudinal

Foam	Density, ρ [kg.m ⁻³]	Type ^a	Modulus of Elasticity, $E \times 10^{-6}$ [Pa]		Proportional limit $\sigma \times 10^{-5}$ [Pa]		Yield ^b strength, $\sigma \times 10^{-5}$ [Pa]		Compressive strength, $\sigma \times 10^{-5}$ [Pa]		Elongation %			
											Plastic		Total	
			T	L	T	L	T	L	T	L	T	L	T	L
KR-1	30,6	PUR	10,1	-	2,2	-	2,6	-	3,2	-	0,88	-	4,01	-
KR-1	30,6	PUR	-	11,2	-	1,04	-	2,1	-	4,0	-	1,98	-	5,69
UR-2	39,2	PUR	11,9	-	2,6	-	4,1	-	6,1	-	0,96	-	6,10	-
UR-2	42,0	PUR	-	15,9	-	4,6	-	6,1	-	6,8	-	0,12	-	4,40
OR-1	47,1	PUR	14,5	-	3,8	-	7,5	-	8,1	-	0,16	-	5,70	-
NR-1	95,8	PUR	38,7	-	6,2	-	10,5	-	17,1	-	1,15	-	5,57	-
NR-1	94,3	PUR	-	45,2	-	6,3	-	11,4	-	17,4	-	1,14	-	4,03
CF-7	32,2	PUF	3,7	-	0,74	-	1,2	-	3,2	-	34,30	-	45,55	-
CF-7	33,6	PUF	-	6,9	-	1,1	-	1,7	-	3,7	-	23,7	-	28,9
YR-1	51,7	PSR	35,0	-	4,5	-	6,8	-	8,8	-	2,81	-	4,91	-
YR-1	50,6	PSR	-	30,2	-	3,2	-	6,4	-	8,7	-	1,83	-	3,92

^a PUR: polyurethane rigid; PUF: polyurethane flexible; PSR: polystyrene rigid.

^b At 0,2% offset.

NOTE COMMENTS: 1) The rigid foam samples were cylindrical, $2,866 \times 10^{-2}$ m in diameter and $7,62 \times 10^{-2}$ m long. Specimens having a reduced middle subclause $6,21 \times 10^{-2}$ m in diameter and $2,54 \times 10^{-2}$ m long m in side and $7,62 \times 10^{-2}$ m long. The specimens were epoxied to polycarbonate grips. Polycarbonate was chosen because its thermal expansion matched that of the adhesive at low temperature.

2) The rate of head movement of the testing machine was 5×10^{-4} m.min⁻¹.

3) Each value represents the average of about four tests. Strength data variation from the reported average were $\pm 10\%$. Suggested inaccuracies of the reported average strength data are $\pm 5\%$. Average elongation data are supposed to be inaccurate to $\pm 5\%$, and data variations from the reported averages are $\pm 10\%$. For a given test individual strength and elongation measurements are thought to be accurate to $\pm 1\%$.

NOTE From Reed, Arvidson & Durcholz (1972) [52].

Table 4-7: Average Tensile Data of Polyurethane and Polystyrene Foams at $T = 300$
KT: Transverse; L: Longitudinal

Foam	Density, ρ [kg.m ⁻³]	Type ^a	Modulus of Elasticity, $E \times 10^{-6}$ [Pa]		Proportional limit $\sigma \times 10^{-5}$ [Pa]		Yield ^b strength, $\sigma \times 10^{-5}$ [Pa]		Compressive strength, $\sigma \times 10^{-5}$ [Pa]		Elongation %			
											Plastic		Total	
			T	L	T	L	T	L	T	L	T	L	T	L
KR-1	30,8	PUR	5,6	-	1,03	-	1,3	-	2,9	-	4,84	-	9,96	-
UR-2	40,0	PUR	16,1	-	1,0	-	2,4	-	5,0	-	2,86	-	5,89	-
UR-2	41,2	PUR	-	13,7	-	1,5	-	2,6	-	4,9	-	2,42	-	6,02
OR-1	47,3	PUR	17,2	-	2,4	-	3,7	-	5,4	-	1,57	-	4,35	-
NR-1	94,7	PUR	26,4	-	3,7	-	6,1	-	10,8	-	3,09	-	7,05	-
NR-1	94,7	PUR	-	31,0	-	3,4	-	6,3	-	10,4	-	2,00	-	5,68
CF-7	33,0	PUF	0,12	-	0,20	-	0,28	-	0,73	-	37,25	-	96,27	-
CF-7	33,2	PUF	-	0,19	-	1,6	-	0,28	-	0,85	-	35,33	-	82,03
YR-1	50,8	PSR	36,1	-	3,0	-	6,2	-	8,2	-	2,84	-	5,24	-
YR-1	50,9	PSR	-	38,7	-	3,9	-	6,3	-	8,3	-	2,10	-	4,20

^a PUR: polyurethane rigid; PUF: polyurethane flexible; PSR: polystyrene rigid.

^b At 0,2% offset.

NOTE COMMENTS: 1) The rigid foam samples were cylindrical, $2,866 \times 10^{-2}$ m in diameter and $7,62 \times 10^{-2}$ m long. Specimens having a reduced middle subclause $6,21 \times 10^{-2}$ m in diameter and $2,54 \times 10^{-2}$ m long m in side and $7,62 \times 10^{-2}$ m long. The specimens were epoxied to polycarbonate grips. Polycarbonate was chosen because its thermal expansion matched that of the adhesive at low temperature.
 2) The rate of head movement of the testing machine was 5×10^{-4} m.min⁻¹.
 3) Each value represents the average of about four tests. Strength data variation from the reported average were $\pm 10\%$. Suggested inaccuracies of the reported average strength data are $\pm 5\%$. Average elongation data are supposed to be inaccurate to $\pm 5\%$, and data variations from the reported averages are $\pm 10\%$. For a given test individual strength and elongation measurements are thought to be accurate to $\pm 1\%$.

NOTE From Reed, Arvidson & Durcholz (1972) [52].

Table 4-8: Average Compressive Data of Polyurethane and Polystyrene Foams

Data at $T = 76$ K	Table 4-9
Data at $T = 195$ K	Table 4-10
Data at $T = 300$ K	Table 4-11

NOTE Individual strength and elongation measurements are accurate to $\pm 1\%$. Reported strength and elongation measurements are thought to be accurate to $\pm 7\%$, with the principal contributor to test inaccuracy being material inconsistency.

NOTE From Arvidson, Durcholz & Reed (1972) [4].

Table 4-9: Average Compressive Data of Polyurethane and Polystyrene Foams at $T = 76\text{ K}$ T: Transverse; L: Longitudinal

Foam	Density, ρ [kg.m ⁻³]	Type ^a	Modulus of Elasticity, $E \times 10^{-6}$ [Pa]		Proportional limit $\sigma \times 10^{-5}$ [Pa]		Yield strength, $\sigma \times 10^{-5}$ [Pa]		Compressive Strength ^c , $\sigma \times 10^{-5}$ [Pa]		Elongation %			
											Plastic		Total	
			T	L	T	L	T	L	T	L	T	L	T	L
UR-2	47,6	PUR	29,4	-	4,0	-	5,1	-	6,3	-	0,41	-	2,58	-
PR-1	46,8	PUR	-	36,1	-	5,4	-	-	-	6,5	-	0,11	-	1,92
NR-1	48,4	PUR	13,6	-	2,3	-	3,8	-	6,2	-	2,69	-	7,32	-
CF-7	47,4	PUF	-	23,0	-	3,5	-	5,8	-	7,0	-	0,72	-	3,75
YR-1	94,8	PSR	67,8	-	11,3	-	-	-	14,8	-	0,04	-	2,39	-
ZR-1	95,6	PSR	-	81,7	-	10,8	-	-	-	14,2	-	0,13	-	1,87
	34,1		12,0	-	2,1	-	2,8	-	3,0	-	0,35	-	2,88	-
	31,7		-	10,9	-	2,2	-	3,2	-	3,4	-	0,45	-	3,55
	51,6		31,6	-	3,7	-	5,0	-	5,4	-	3,28	-	5,02	-
	51,3		-	32,4	-	3,7	-	4,6	-	5,1	-	4,65	-	6,28
	99,5		73,8	-	9,3	-	13,7	-	15,1	-	0,58	-	2,32	-
	99,2		-	74,8	-	9,4	-	13,2	-	14,5	-	0,95	-	2,93

^a PUR: polyurethane rigid; PUF: polyurethane flexible; PSR: polystyrene rigid.

^b At 0,2% offset.

^c Tests in which the stress continued to increase as the test progressed (i.e., the material did not exhibit an ultimate) were concluded at 10% total elongation. The corresponding values are in parentheses.

NOTE COMMENTS: 1) The rigid foam samples were cylindrical, $2,866 \times 10^{-2}$ m in diameter and $2,54 \times 10^{-2}$ m long. Flexible specimens were machined into a cube geometry with the length of each side equal to $2,54 \times 10^{-2}$ m.

2) The rate of head movement of the testing machine was 5×10^{-4} m.min⁻¹.

3) Each value represents the average of at least three tests. The average strength deviations for a given type and loading direction were $\pm 9\%$, while the average elongation deviations were $\pm 8\%$. Individual strength and elongation measurements are accurate to $\pm 1\%$. Reported strength and elongation measurements are thought to be accurate to $\pm 7\%$, with the principal contributor to test inaccuracy being material inconsistency.

NOTE From Arvidson, Durcholz & Reed (1972) [4].

Table 4-10: Average Compressive Data of Polyurethane and Polystyrene Foams at
 $T = 195 \text{ K}$ T: Transverse; L: Longitudinal

Foam	Density, ρ [kg.m ⁻³]	Type ^a	Modulus of Elasticity, $E \times 10^{-6}$ [Pa]		Proportional limit $\sigma \times 10^{-5}$ [Pa]		Yield ^b strength, $\sigma \times 10^{-5}$ [Pa]		Compressive Strength ^c , $\sigma \times 10^{-5}$ [Pa]		Elongation %			
											Plastic		Total	
			T	L	T	L	T	L	T	L	T	L	T	L
UR-2	43,9	PUR	12,6	-	1,7	-	2,6	-	3,8	-	1,84	-	4,91	-
NR-1	41,3	PUR	-	11,2	-	1,7	-	2,3	-	3,4	-	2,35	-	5,59
CF-7	94,2	PUF	40,4	-	4,1	-	7,0	-	(11.6)	-	7,10	-	10,00	-
YR-1	94,5	PSR	-	51,4	-	4,6	-	7,3	-	(11.7)	-	7,68	-	10,00-
	33,3		2,0	-	0,33	-	0,57	-	(0.84)	-	5,96	-	10,00	5,43
	32,0		-	3,5	-	0,72	-	0,99	-	1,19	-	1,87	-	-
	51,3		23,6	-	3,2	-	4,1	-	(5.2)	-	7,80	-	10,00	10,00
	51,4		-	21,5	-	3,0	-	3,8	-	(4.8)	-	7,70	-	-

^a PUR: polyurethane rigid; PUF: polyurethane flexible; PSR: polystyrene rigid.

^b At 0,2% offset.

^c Tests in which the stress continued to increase as the test progressed (i.e., the material did not exhibit an ultimate) were concluded at 10% total elongation. The corresponding values are in parentheses.

NOTE COMMENTS: 1) The rigid foam samples were cylindrical, $2,866 \times 10^{-2}$ m in diameter and $2,54 \times 10^{-2}$ m long. Flexible specimens were machined into a cube geometry with the length of each side equal to $2,54 \times 10^{-2}$ m.
 2) The rate of head movement of the testing machine was 5×10^{-4} m.min⁻¹.
 3) Each value represents the average of at least three tests. The average strength deviations for a given type and loading direction were $\pm 9\%$, while the average elongation deviations were $\pm 8\%$. Individual strength and elongation measurements are accurate to $\pm 1\%$. Reported strength and elongation measurements are thought to be accurate to $\pm 7\%$, with the principal contributor to test inaccuracy being material inconsistency.

NOTE From Arvidson, Durcholz & Reed (1972) [4].

Table 4-11: Average Compressive Data of Polyurethane and Polystyrene Foams at $T = 300\text{ K}$
T: Transverse; L: Longitudinal

Foam	Density, ρ [kg.m ⁻³]	Type ^a	Modulus of Elasticity, $E \times 10^{-6}$ [Pa]		Proportional limit $\sigma \times 10^{-5}$ [Pa]		Yield strength, $\sigma \times 10^{-5}$ [Pa]		Compressive strength ^c , $\sigma \times 10^{-5}$ [Pa]		Elongation %			
											Plastic		Total	
			T	L	T	L	T	L	T	L	T	L	T	L
UR-2	46,3	PUR	7,0	-	1,3	-	1,9	-	3,0	-	3,26	-	7,62	-
NR-1	40,8	PUR	-	7,2	-	1,4	-	2,2	-	3,0	-	2,04	-	6,34
CF-7	95,1	PUF	17,2	-	3,5	-	5,2	-	8,0	-	3,27	-	7,93	-
YR-1	93,2	PSR	-	20,2	-	4,3	-	5,9	-	8,1	-	3,27	-	7,38
	33,3		0,06	-	0,021	-	0,024	-	(0.035)	-	4,13	-	10,00	-
	33,3		-	0,20	-	0,014	-	0,021	-	(0.04)	-	7,31	-	10,00
	51,9		19,8	-	3,4	-	3,7	-	(4.0)	-	7,94	-	10,00	-
	51,1		-	19,1	-	2,3	-	3,2	-	3,4	-	0,68	-	2,46

^a PUR: polyurethane rigid; PUF: polyurethane flexible; PSR: polystyrene rigid.

^b At 0,2% offset.

^c Tests in which the stress continued to increase as the test progressed (i.e., the material did not exhibit an ultimate) were concluded at 10% total elongation. The corresponding values are in parentheses.

NOTE COMMENTS: 1) The rigid foam samples were cylindrical, $2,866 \times 10^{-2}$ m in diameter and $2,54 \times 10^{-2}$ m long. Flexible specimens were machined into a cube geometry with the length of each side equal to $2,54 \times 10^{-2}$ m.
 2) The rate of head movement of the testing machine was 5×10^{-4} m.min⁻¹.
 3) Each value represents the average of at least three tests. The average strength deviations for a given type and loading direction were $\pm 9\%$, while the average elongation deviations were $\pm 8\%$. Individual strength and elongation measurements are accurate to $\pm 1\%$. Reported strength and elongation measurements are thought to be accurate to $\pm 7\%$, with the principal contributor to test inaccuracy being material inconsistency.

NOTE From Arvidson, Durcholz & Reed (1972) [4].

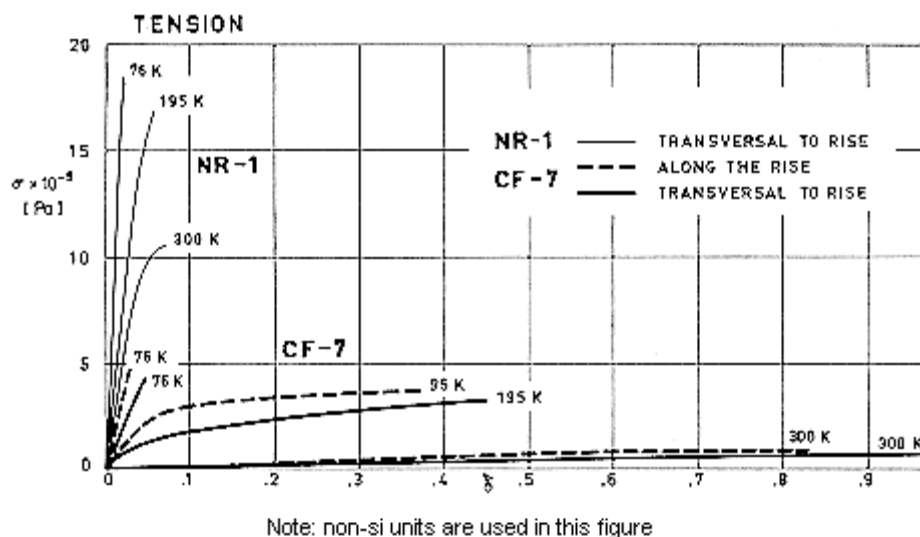


Figure 4-13: Tensile stress, σ , vs. strain, δ , for several polyurethane foams at 76, 195 and 300 K.

From Reed, Arvidson & Durcholz (1972) [52].

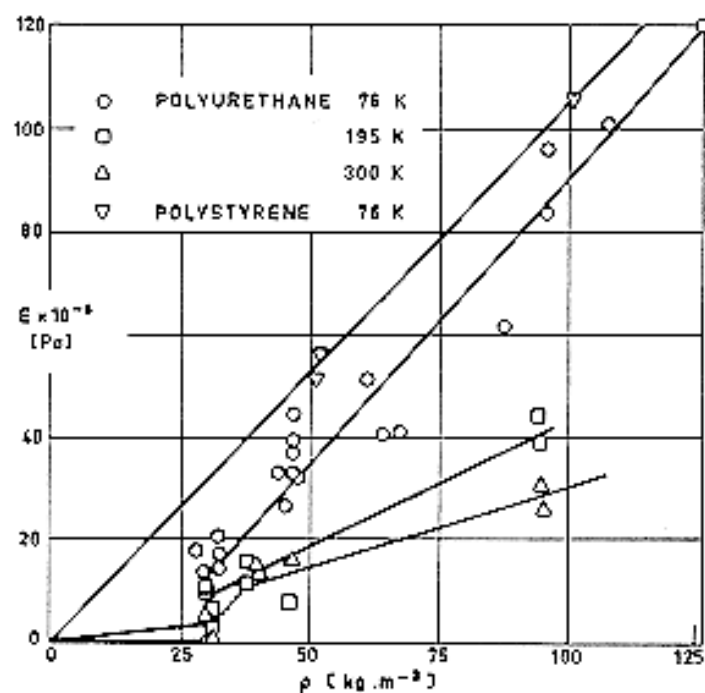


Figure 4-14: Modulus of Elasticity-tensile- E , as a function of density, ρ , for several organic foams.

Solid lines show the estimated dependence of E on foam density; dashed lines indicate transition between rigid foams (large E) and flexible foams (small E). From Reed et al. (1972) [52].

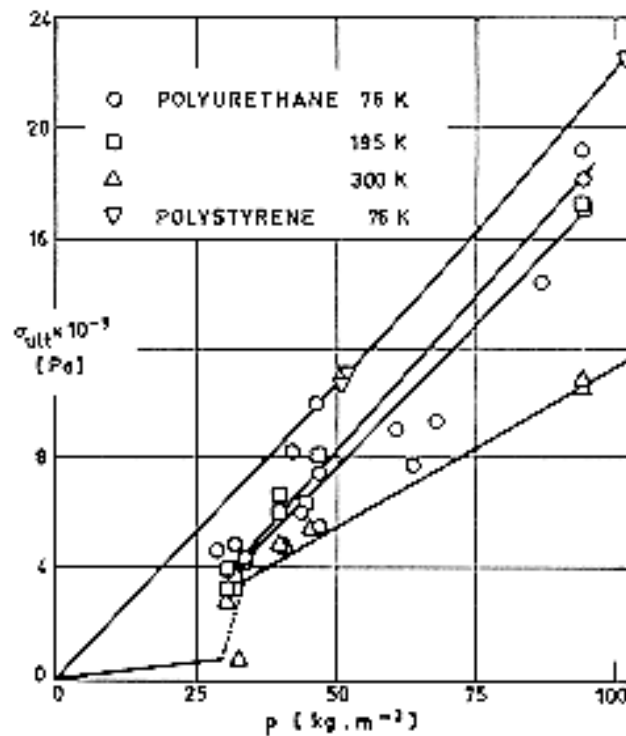


Figure 4-15: Ultimate tensile strength, σ_{ult} , as a function of density, ρ , for several organic foams.

Solid and dashed lines have the same meaning as in Figure 4-8. From Reed et al. (1972) [52].

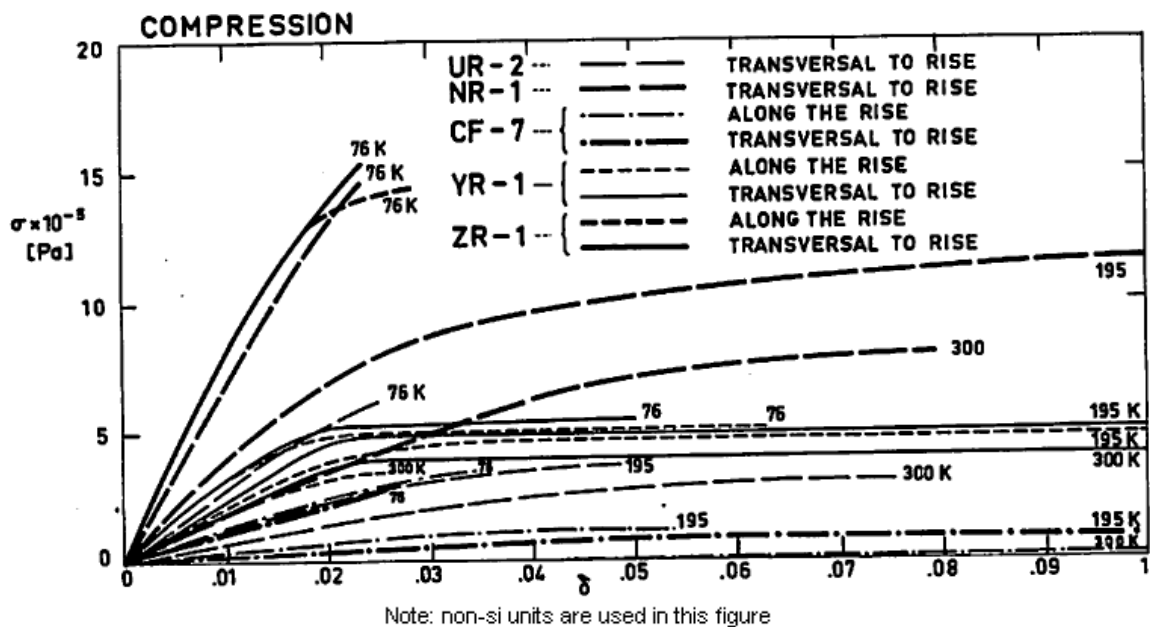


Figure 4-16: Compressive stress, s , vs, strain, d , for several organic foams at 76, 195 and 300 K.

From Arvidson, Durcholz & Reed (1972) [4].

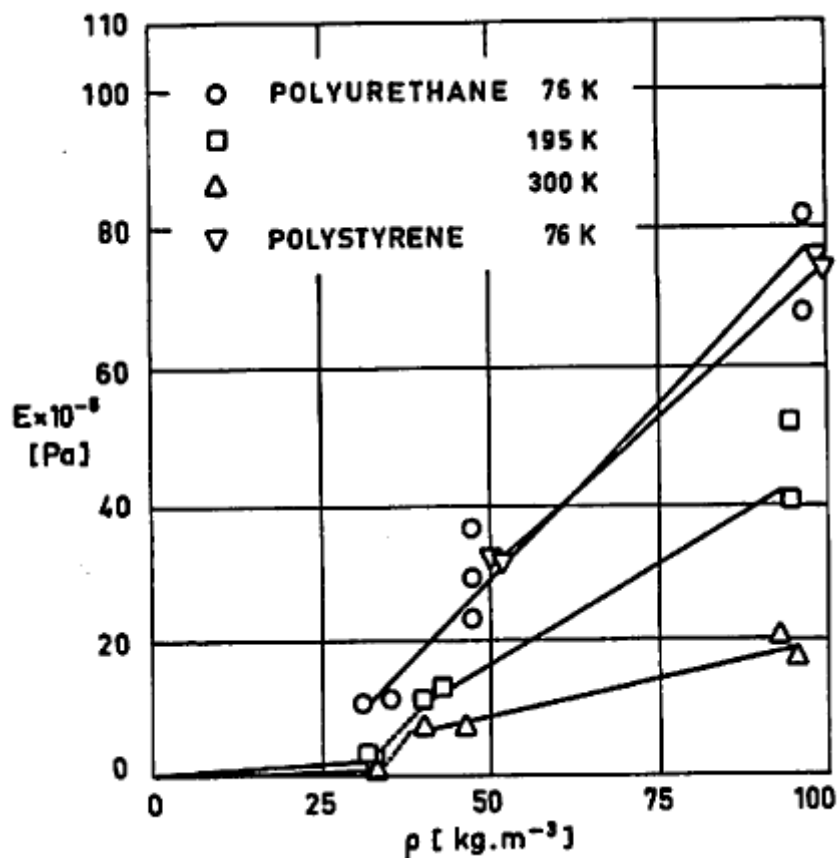


Figure 4-17: Modulus of Elasticity-tensile- E , as a function of density, ρ , for several organic foams.

Solid lines show the estimated dependence of E on foam density; dashed lines indicate transition between rigid foams (large E) and flexible foams (small E). From Arvidson et al. (1972) [4].